



Cr被覆型事故耐性燃料の開発

原子力システム研究開発事業における事故耐性燃料の開発

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福島第一原子力発電所事故の教訓を元に、事故時の事象進展を抑制する事故耐性燃料が世界的に注目されており、複数の燃料概念が提案されている。本事業では、短期的に最も実現性が高いとされる金属被覆ジルカロイ合金に着目し、その開発指針の構築と最適な材料の開発を目的とする。



事故耐性燃料

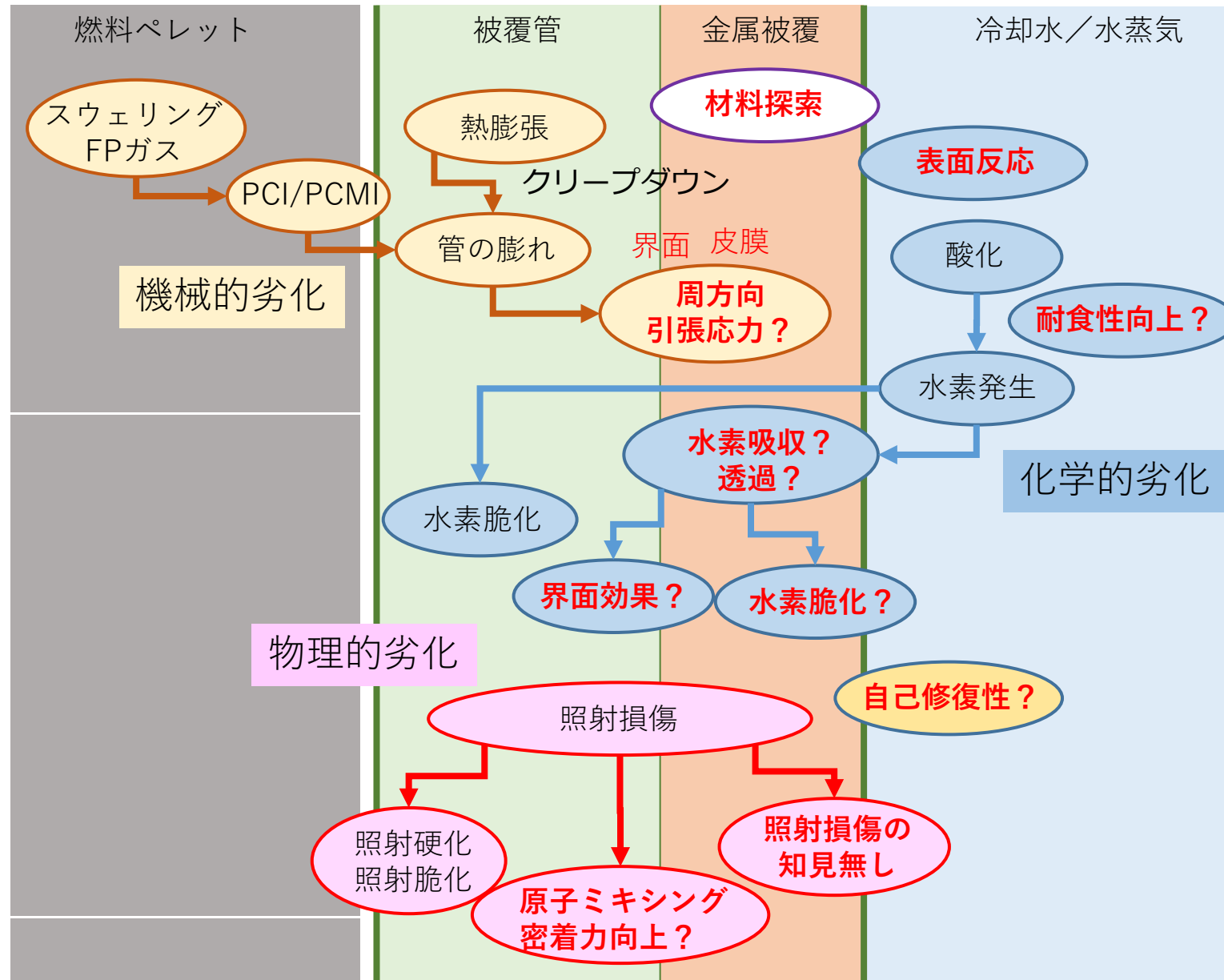
OECD/NEA、原子力学会炉心燃料分科会等における検討

	事故耐性	技術レベル	実現時期	技術レベル (TRL)	
				・性能実証 (TRL7-9)	(→高)
・改良ジルカロイ	中	高	短期	・原理実証 (TRL4-6)	(→中)
・Cr-coated Zr合金	中	初～高	短期	・概念実証 (TRL1-3)	(→初)
・Zr-coated Mo合金	?	初	中長期?	OECD/NEA	
・FeAlCr合金、ODS鋼	高	初～中	中期		
・SiC複合材料	非常に高	初	長期		

Cr被覆Zr合金は、高温での他部材との反応を抑制する思想を取り入れたものであり、Zr合金が実用材であることから、早期の実機導入可能性は最も高い、とされている。



金属被覆ジルカロイの開発と事故耐性確認に必要な科学的知見



原子炉環境における被覆材の特性は良く分かっていない。

Cr酸化膜

- ・ 安定性
- ・ 水素障壁機能
- ・ 自己修復性

被覆

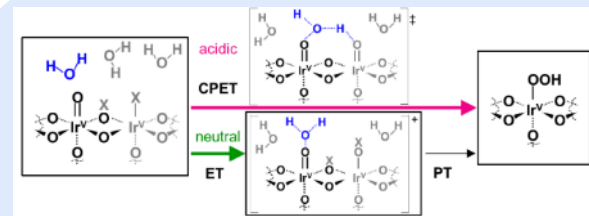
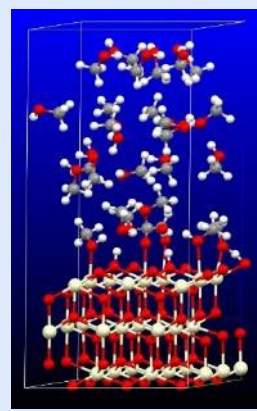
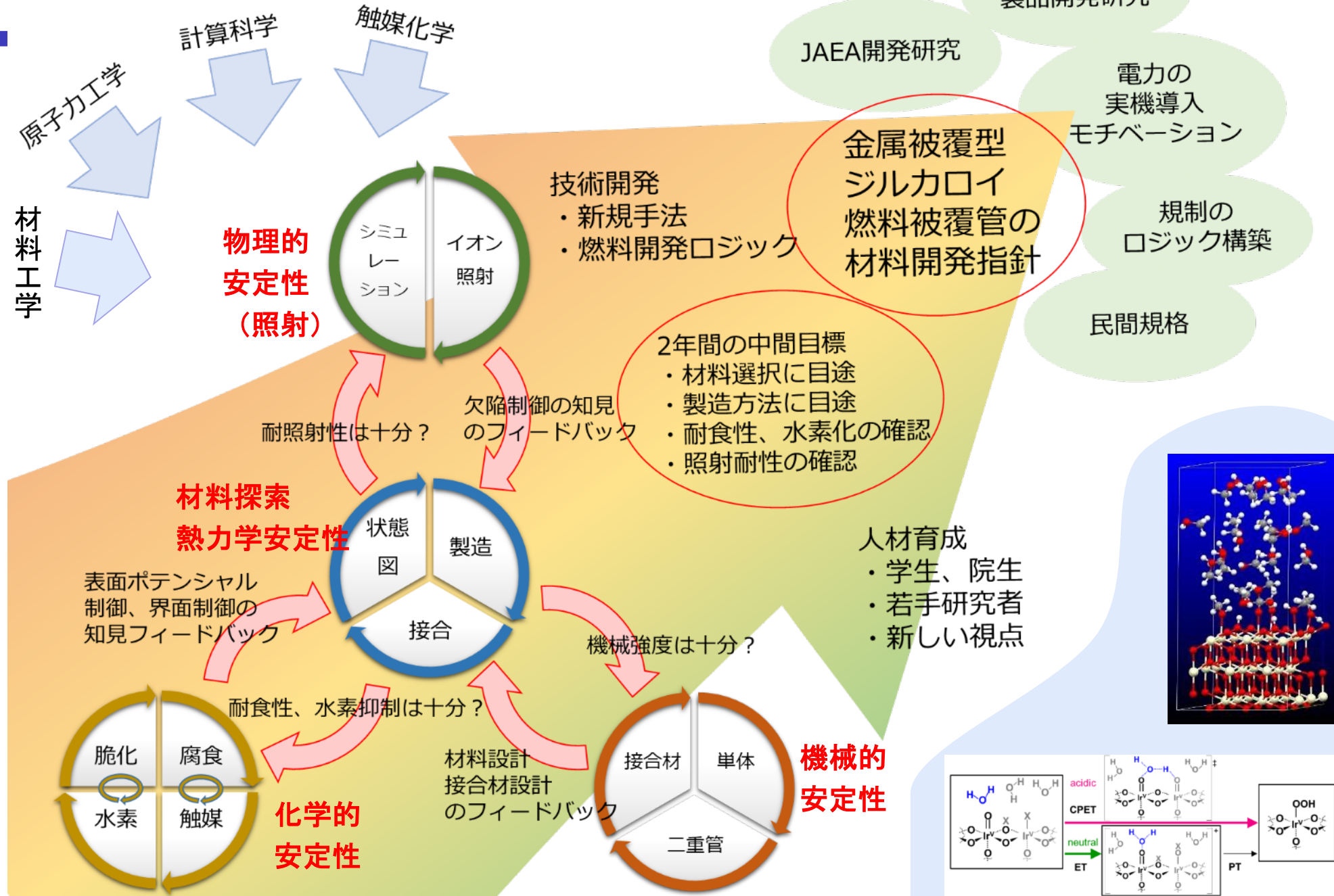
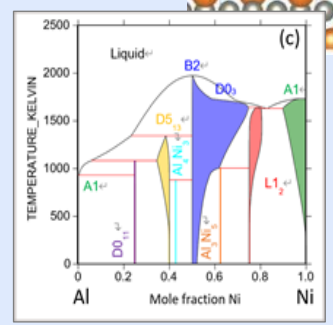
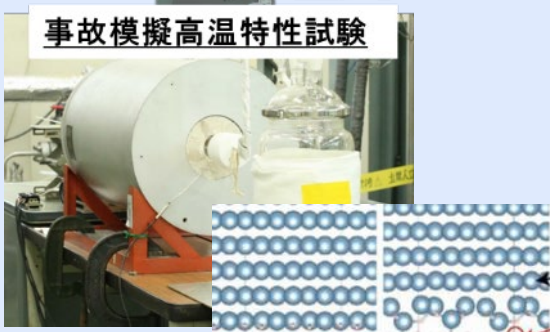
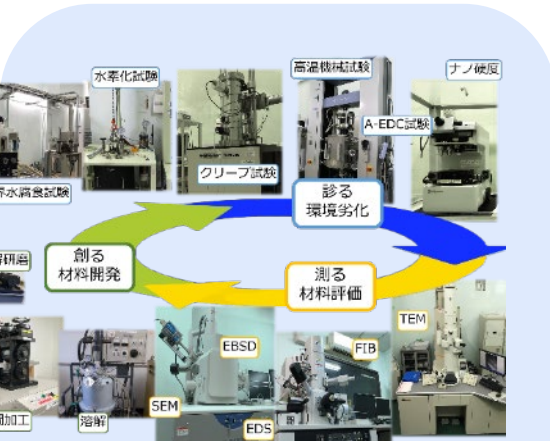
- ・ 安定性

界面

- ・ 安定性
- ・ 水素との作用



研究の体系





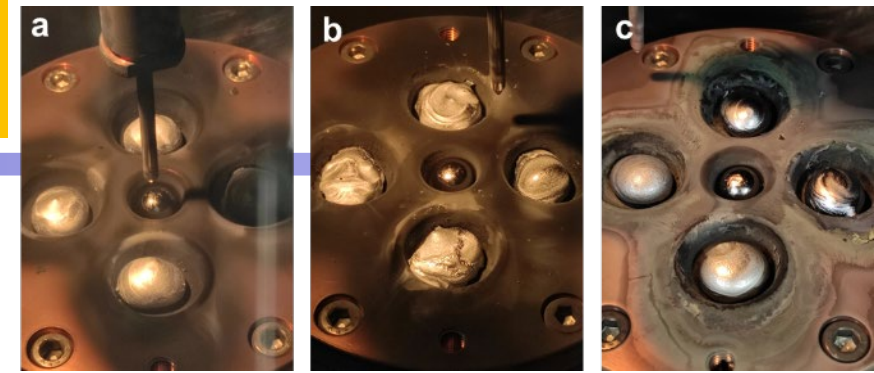
材料開発



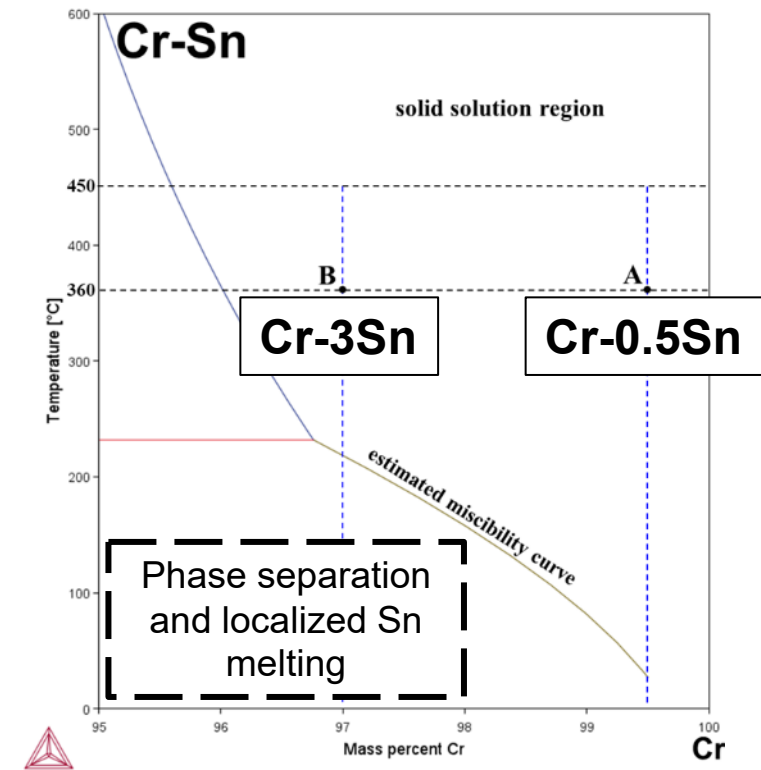
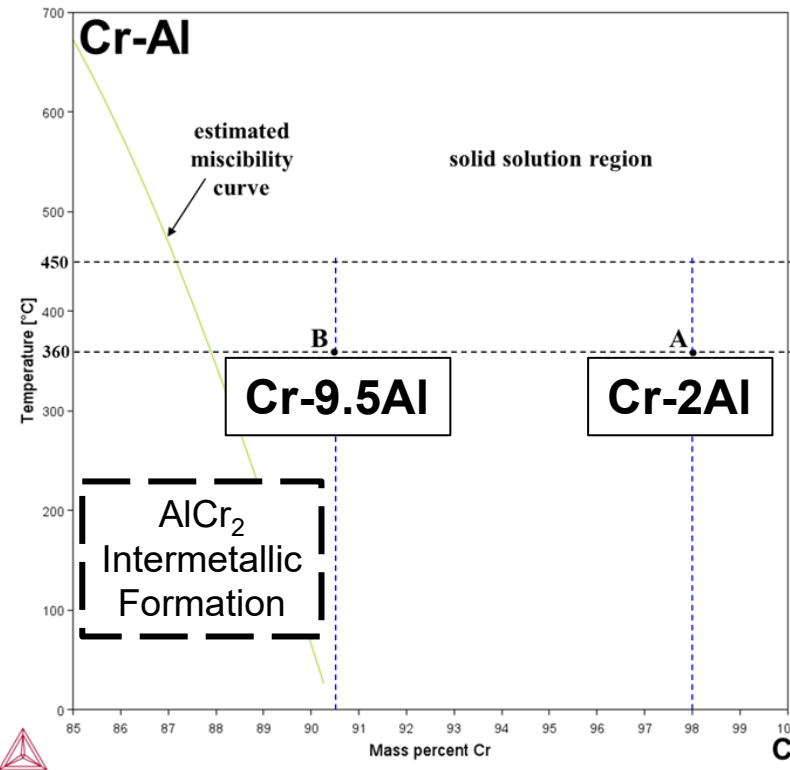
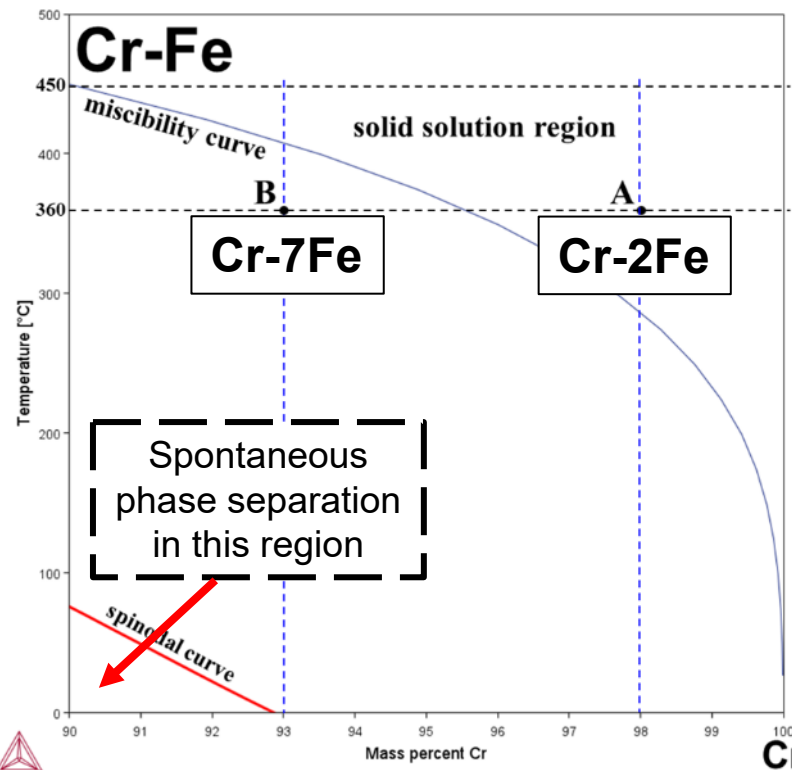
Selection of Alloying Elements

- ❑ Low σ satisfies neutronic requirements
- ❑ Higher k indicates good heat transfer

Element	$\sigma_{\text{thermal neutron (b)}}$ [1]	k (W/m-K) [2]
Sn	~ 6.5 (^{120}Sn)	< 60
Zr	~ 6 (^{90}Zr)	21
Fe	~ 4.7 (^{56}Fe)	55
Cr	~ 4.3 (^{52}Cr)	81
Al	~ 1.87 (^{27}Al)	232



- ❑ Prevent the spinodal decomposition in Cr-Fe
- ❑ Inhibit intermetallic formation in Cr-Al
- ❑ Avoid phase separation in Cr-Sn



[1] N. Soppera, E. Dupont, M. Fleming "JANIS Book of neutron-induced cross-sections" OECD NEA Data Bank (2020).

[2] R. H. Perry, D. W. Green "Perry's chemical engineers' handbook" New York: McGraw-Hill (2008).



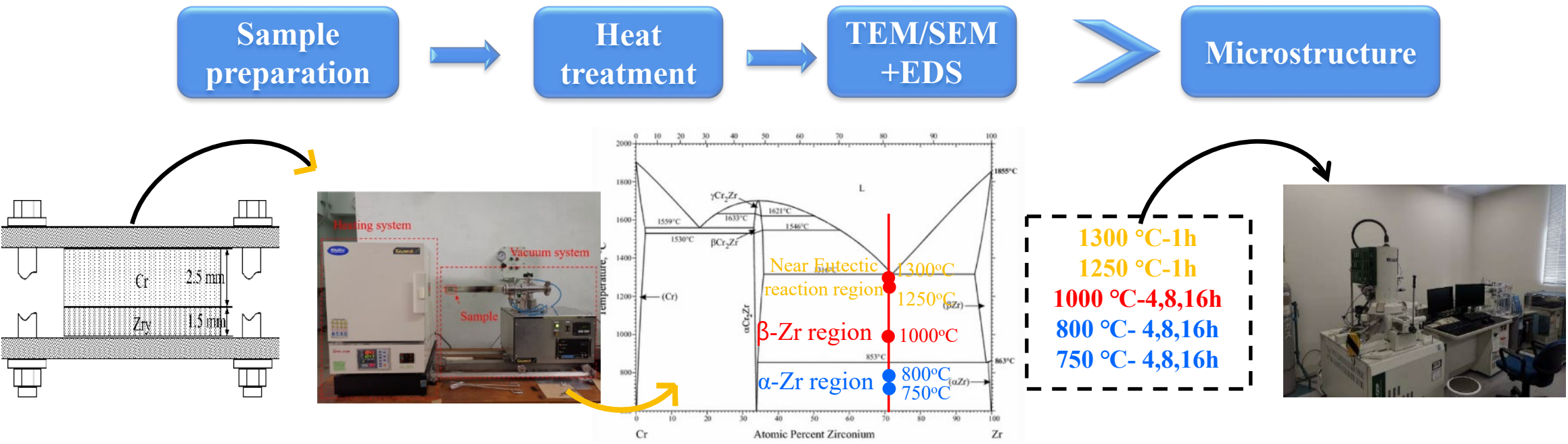
Diffusion bonding of Cr and Zry4

➤ Materials

Chemical Composition of different materials (wt.%)

Materials	Cr	Ni	Mn	Si	Mo	Fe	Sn	Zr
Cr	99.9	-	-	-	-	-	-	-
Zry-4	0.07~0.13	-	-	-	-	0.18~0.24	1.2~1.7	Balance

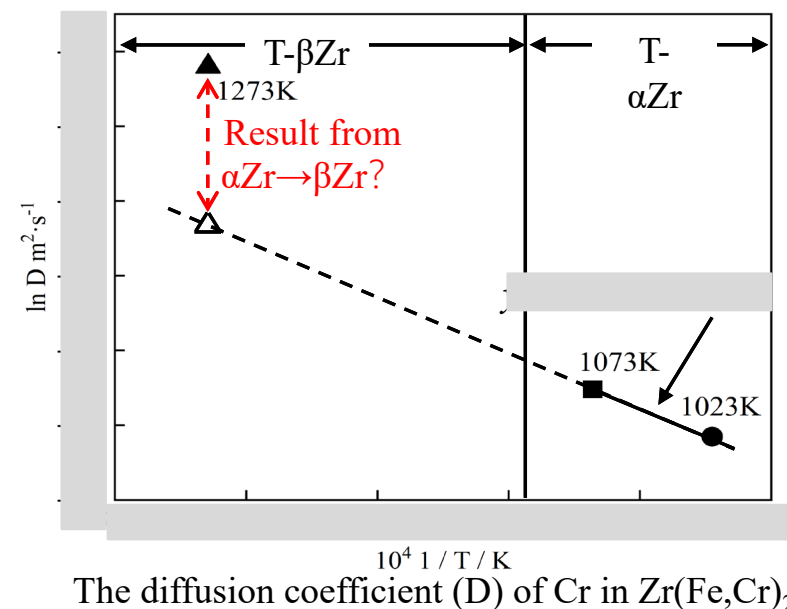
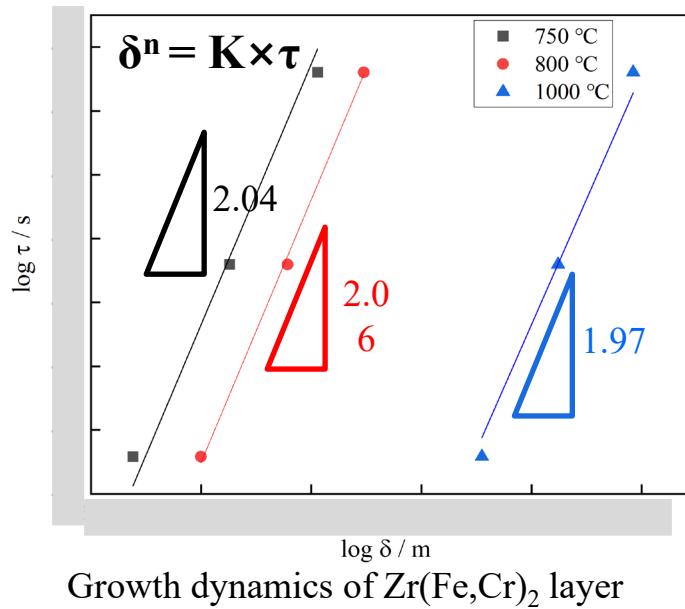
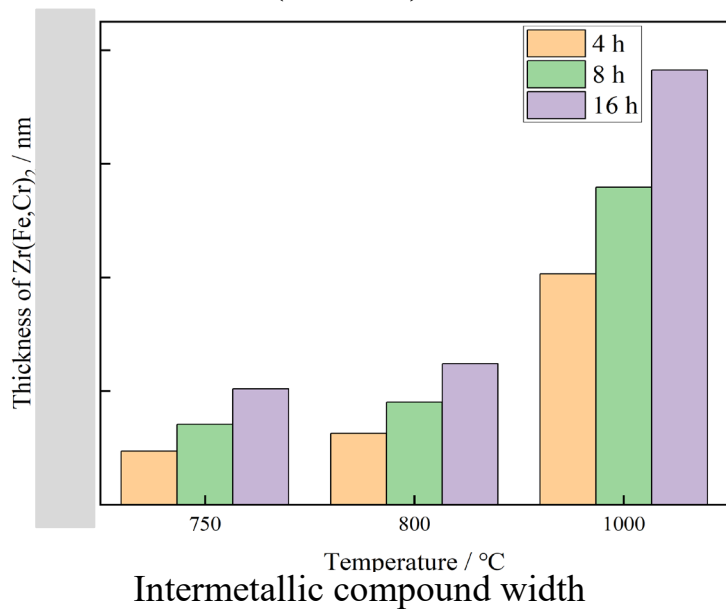
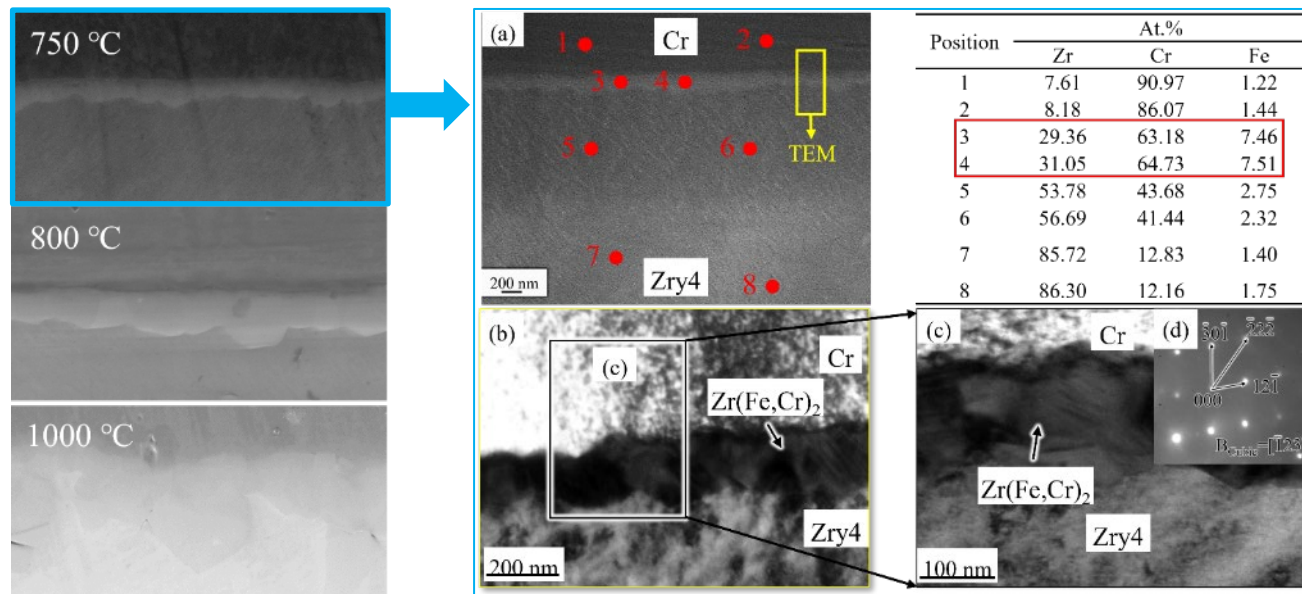
➤ Methods





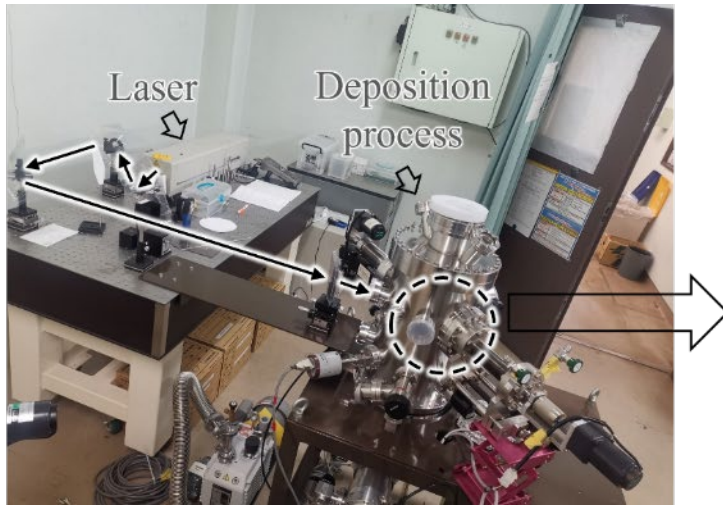
Diffusion bonding of Cr and Zry4

- ✓ The presence of a new layer in the Cr-Zry4 interface region was found under different diffusion conditions.
- ✓ The intermetallic compound Zr(Fe,Cr)_2 produced by diffusion was determined by compositional analysis (EDS) and crystal structure (SEAD).

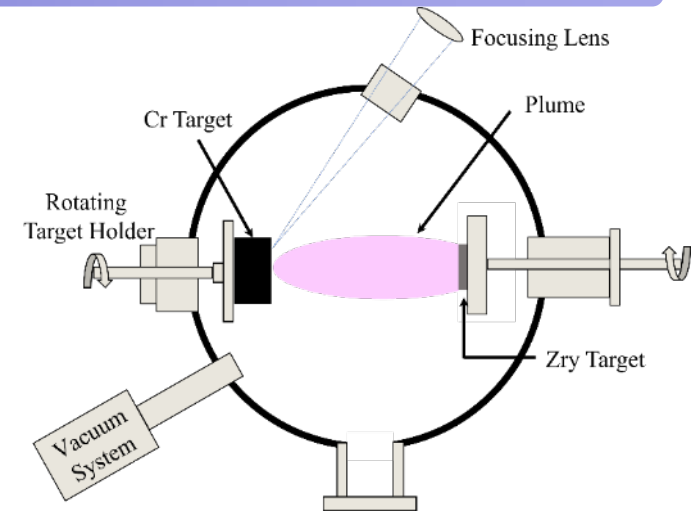
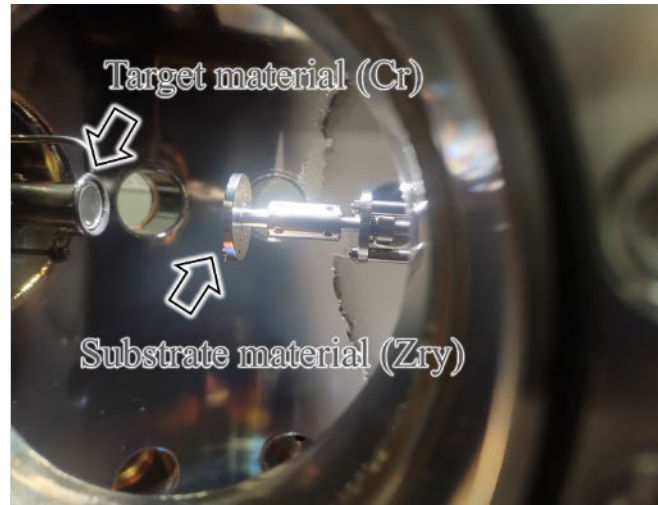




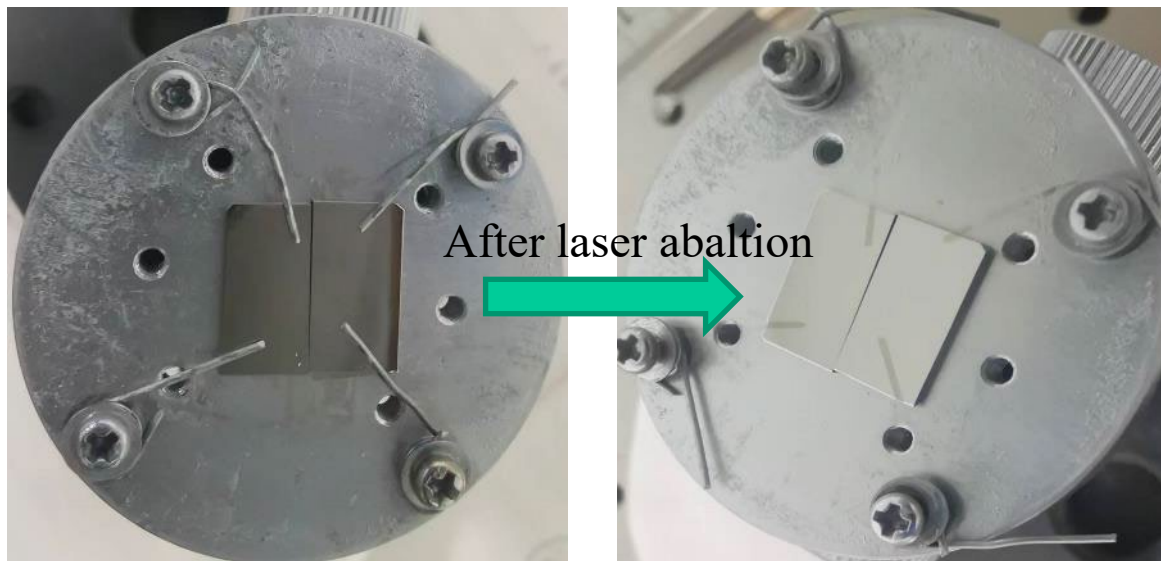
Cr-coated Zry4 by laser ablation



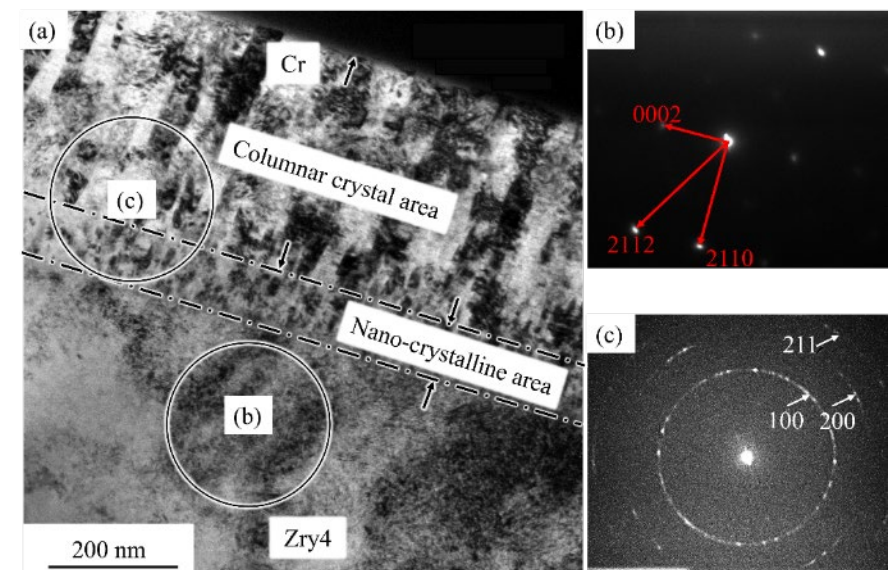
Laser ablation equipment



Schematic diagram of the laser ablation



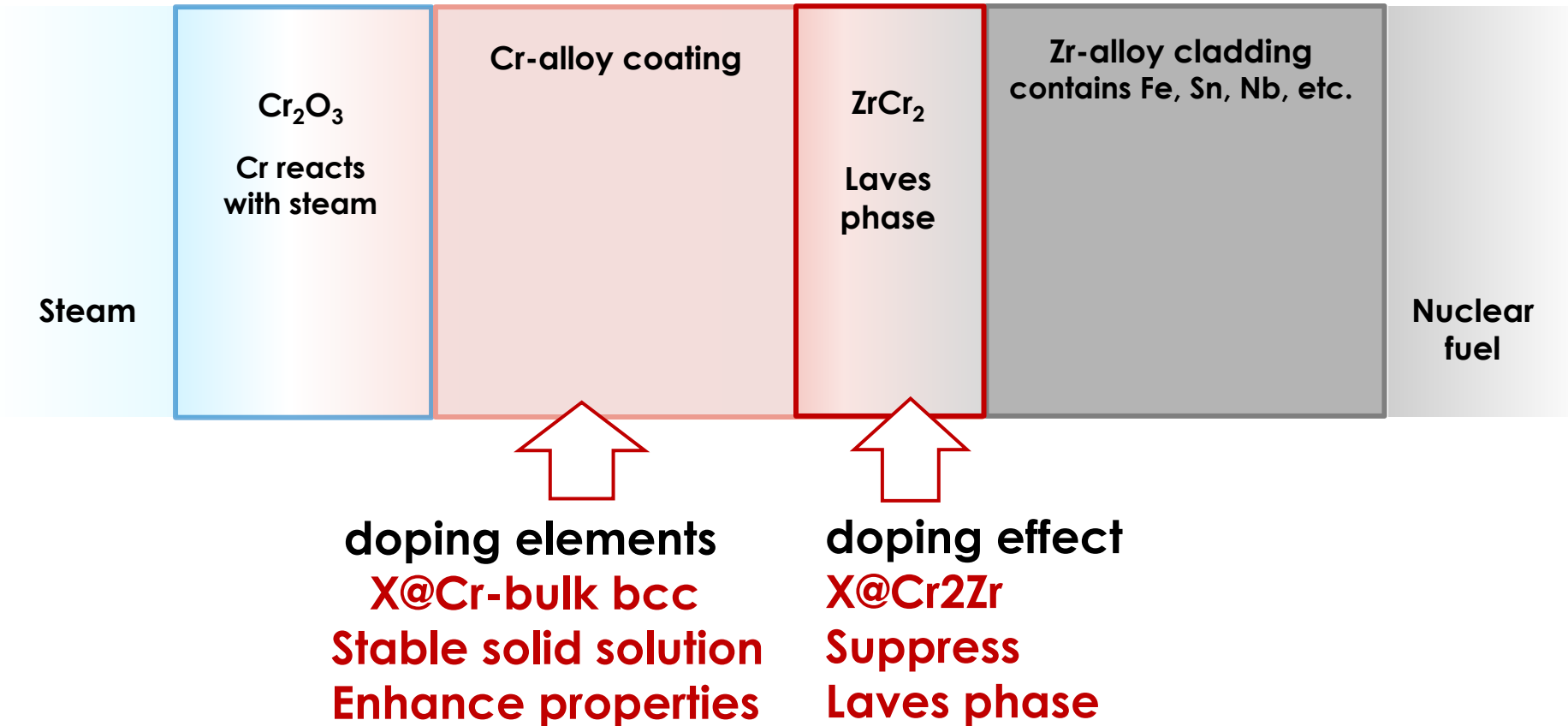
Cr-coated Zry4 by laser ablation



Microstructure of Cr coating layer and substrate

第一原理計算による材料開発

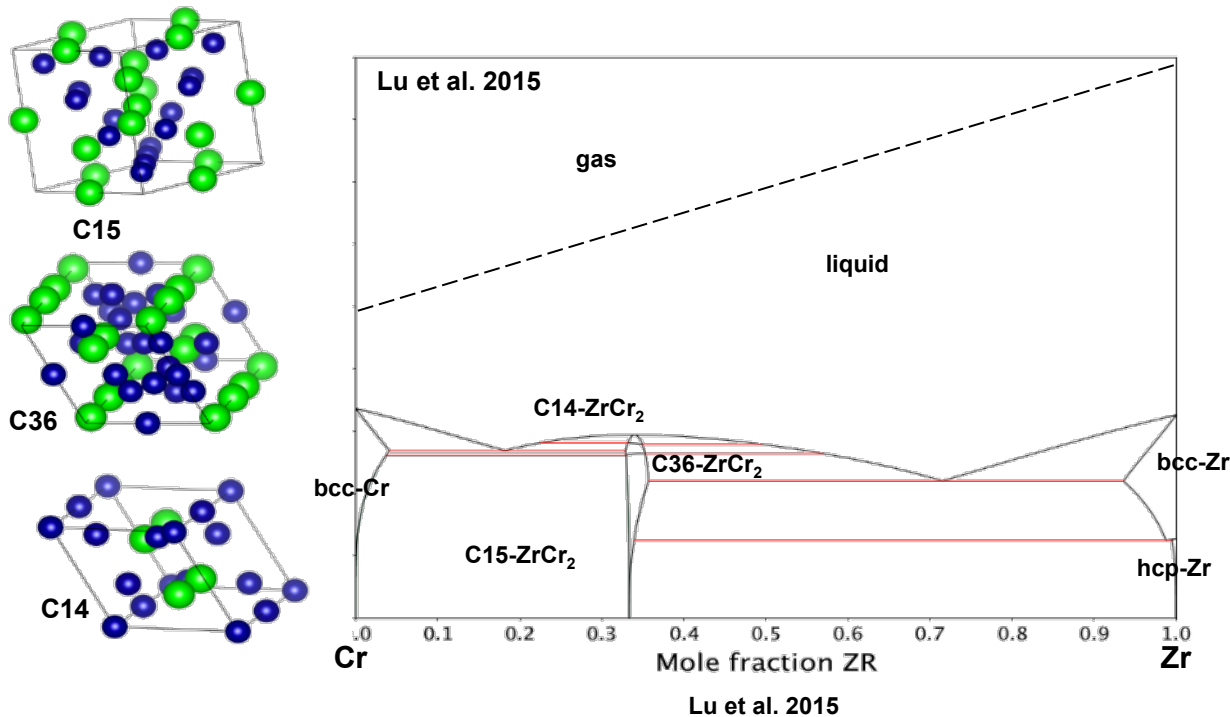
Scenario



Thermodynamic stability, Solubility, Phase diagram

Result-1 X@Zr-Cr Destabilizing Laves Phase ZrCr_2

Exploring proper doping element X@ZrCr_2
at Cr-Zr interface to **destabilize Laves phase**



Periodic table of Elements

1 IA H Hydrogen 1.008	2 IIA He Helium 4.0026	3 Li Lithium 6.94	4 Be Beryllium 9.0122	5 B Boron 10.81	6 C Carbon 12.011	7 N Nitrogen 14.007	8 O Oxygen 15.999	9 F Fluorine 18.998	10 Ne Neon 20.180	11 Na Sodium 22.990	12 Mg Magnesium 24.305	13 Al Aluminum 26.982	14 Si Silicon 28.086	15 P Phosphorus 30.974	16 S Sulfur 32.06	17 Cl Chlorine 35.45	18 Ar Argon 39.948	19 K Potassium 39.098	20 Ca Calcium 40.078	21 Sc Scandium 44.956	22 Ti Titanium 47.88	23 V Vanadium 50.942	24 Cr Chromium 51.996	25 Mn Manganese 54.938	26 Fe Iron 55.845	27 Co Cobalt 58.933	28 Ni Nickel 58.693	29 Cu Copper 63.546	30 Zn Zinc 65.38	31 Ga Gallium 69.723	32 Ge Germanium 72.631	33 As Arsenic 74.922	34 Se Selenium 78.96	35 Br Bromine 79.904	36 Kr Krypton 83.798	37 Rb Rubidium 85.468	38 Sr Strontium 87.62	39 Y Yttrium 88.906	40 Zr Zirconium 91.224	41 Nb Niobium 92.906	42 Mo Molybdenum 95.94	43 Tc Technetium 98	44 Ru Ruthenium 101.07	45 Rh Rhodium 102.906	46 Pd Palladium 106.42	47 Ag Silver 107.868	48 Cd Cadmium 112.411	49 In Indium 114.818	50 Sn Tin 118.710	51 Sb Antimony 121.757	52 Te Tellurium 127.6	53 I Iodine 126.905	54 Xe Xenon 131.29	55 Cs Cesium 132.905	56 Ba Barium 137.327	57-71 Lanthanoids	72 Hf Hafnium 178.49	73 Ta Tantalum 180.948	74 W Tungsten 183.84	75 Re Rhenium 186.207	76 Os Osmium 190.23	77 Ir Iridium 192.222	78 Pt Platinum 195.084	79 Au Gold 196.967	80 Hg Mercury 200.59	81 Tl Thallium 204.38	82 Pb Lead 207.2	83 Bi Bismuth 208.980	84 Po Polonium 209	85 At Astatine 210	86 Rn Radon 222	87 Fr Francium 223	88 Ra Radium 226	89-103 Actinoids	104 Rf Rutherfordium 261	105 Db Dubnium 262	106 Sg Seaborgium 266	107 Bh Bohrium 264	108 Hs Hassium 277	109 Mt Meitnerium 268	110 Ds Darmstadtium 271	111 Rg Roentgenium 272	112 Cn Copernicium 285	113 Nh Nihonium 284	114 Fl Flerovium 289	115 Mc Moscovium 288	116 Lv Livermorium 293	117 Ts Tennessine 294	118 Og Oganesson 294
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Exploring proper alloying elements to
stabilizing/suppressing a structure
via **thermodynamic stability** and
geometric stability.

Formation energy

Formation volume

Results

- Explored stability of doping element X@Cr-Zr
Alloying Zn, Sn, or Mg may suppress formation of the intermetallic Laves phases while controlling volume to prevent cracking.
- Investigated element doping X@Cr-bcc to explore stable **dilute X-Cr solid solution**
- Importance of investigate **competition/combination effect** of doping elements for alloy design.
doping element diffusion trend in 2 interfaces



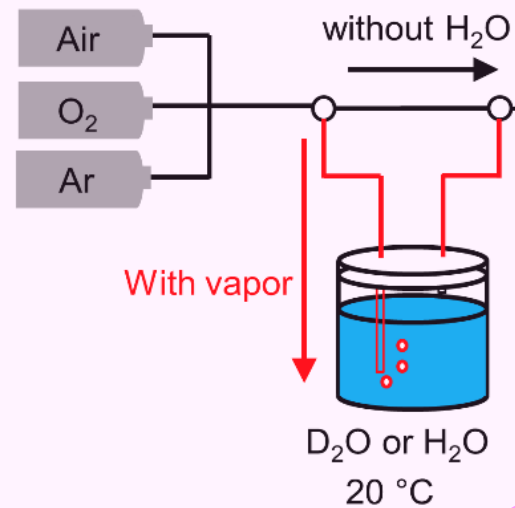
化学的作用



実験の体系

雰囲気 :

- (1) 2.5kPa H₂O / 20kPa O₂ / Ar
- (2) 2.5kPa H₂O / Ar



熱重量分析

分析速度 : 5 K/min

TGA

DRIFTS,
Raman, UV-Vis



質量分析

MS

Reactor

Vent

・ 細断した試料

Zr、Zry、Cr

・ 粉末試料

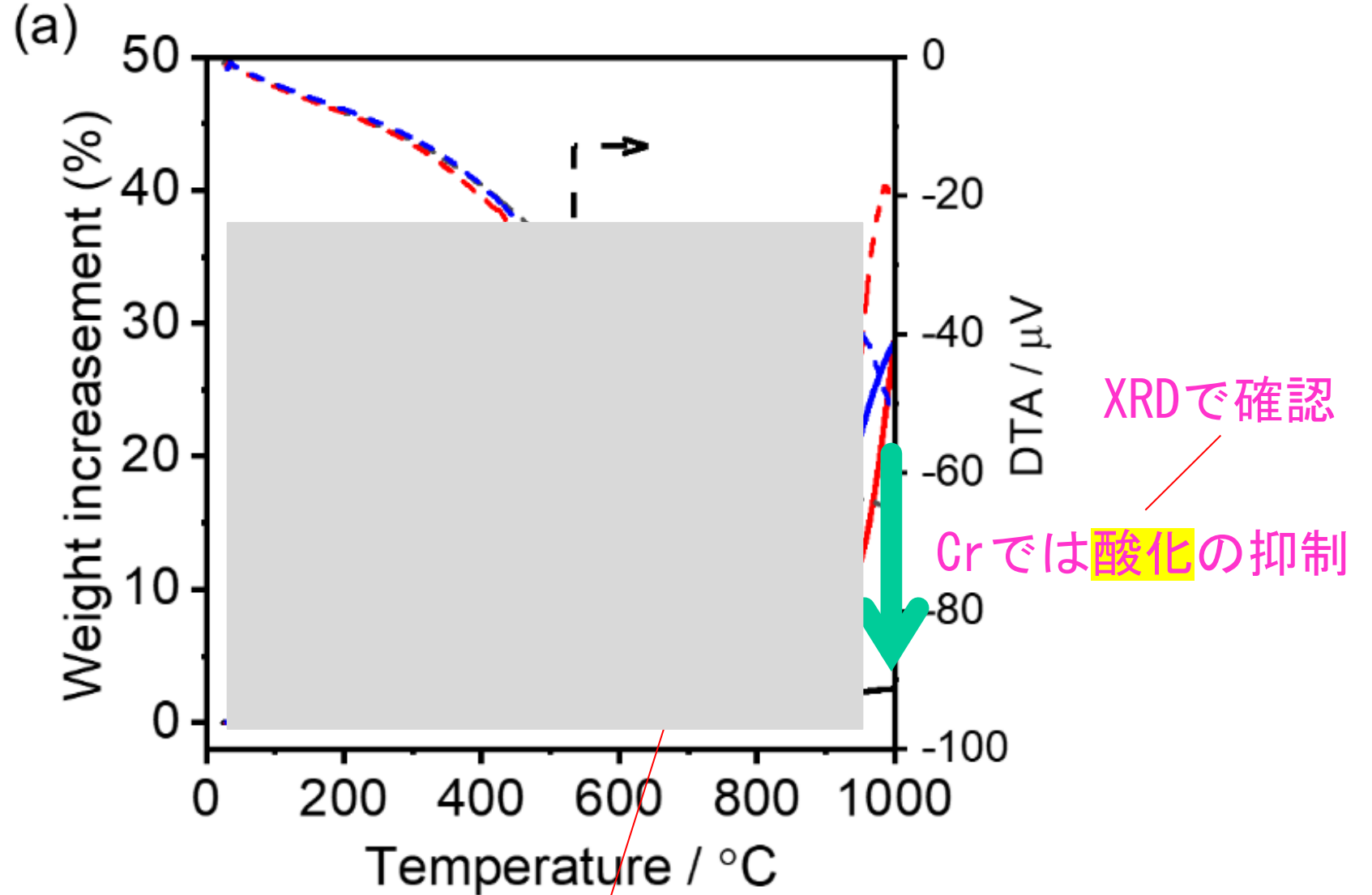
Cr (比表面積一定とし速度論解析)



熱重量分析

雰囲気 :

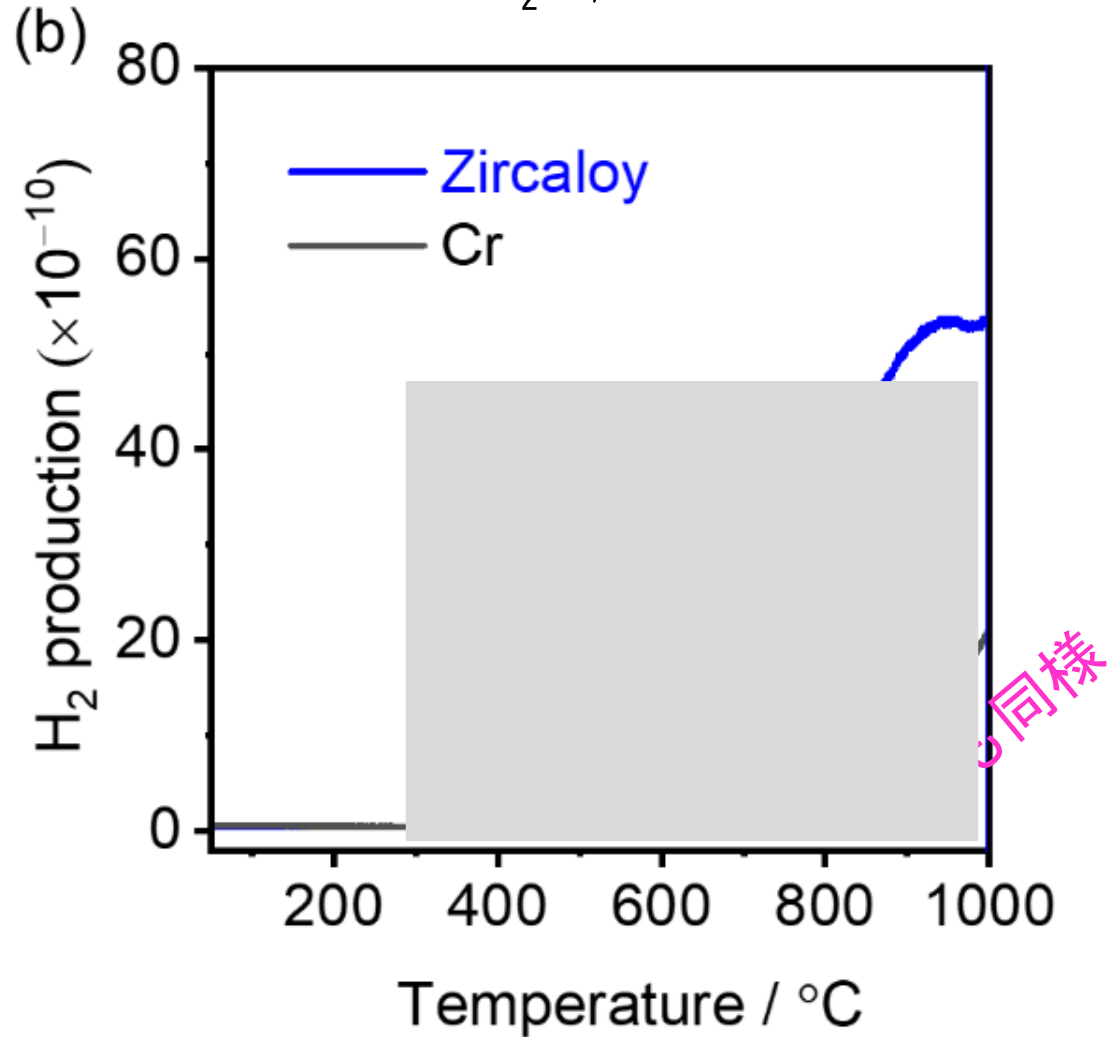
2.5kPa H₂O / 20kPa O₂ / Ar



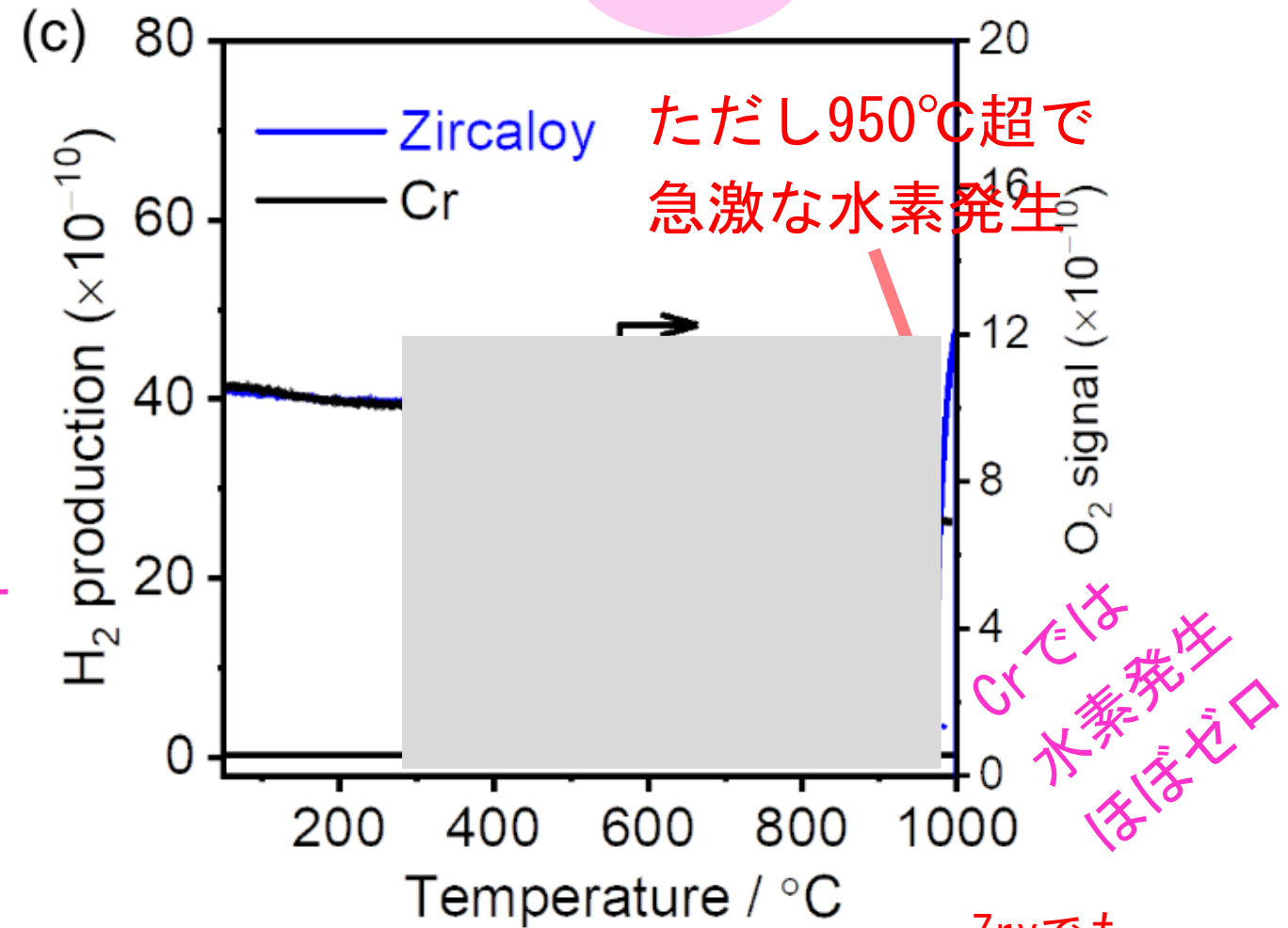


水素の発生とその抑制因子

2.5kPa H₂O / Ar



2.5kPa H₂O / 20kPa O₂ / Ar

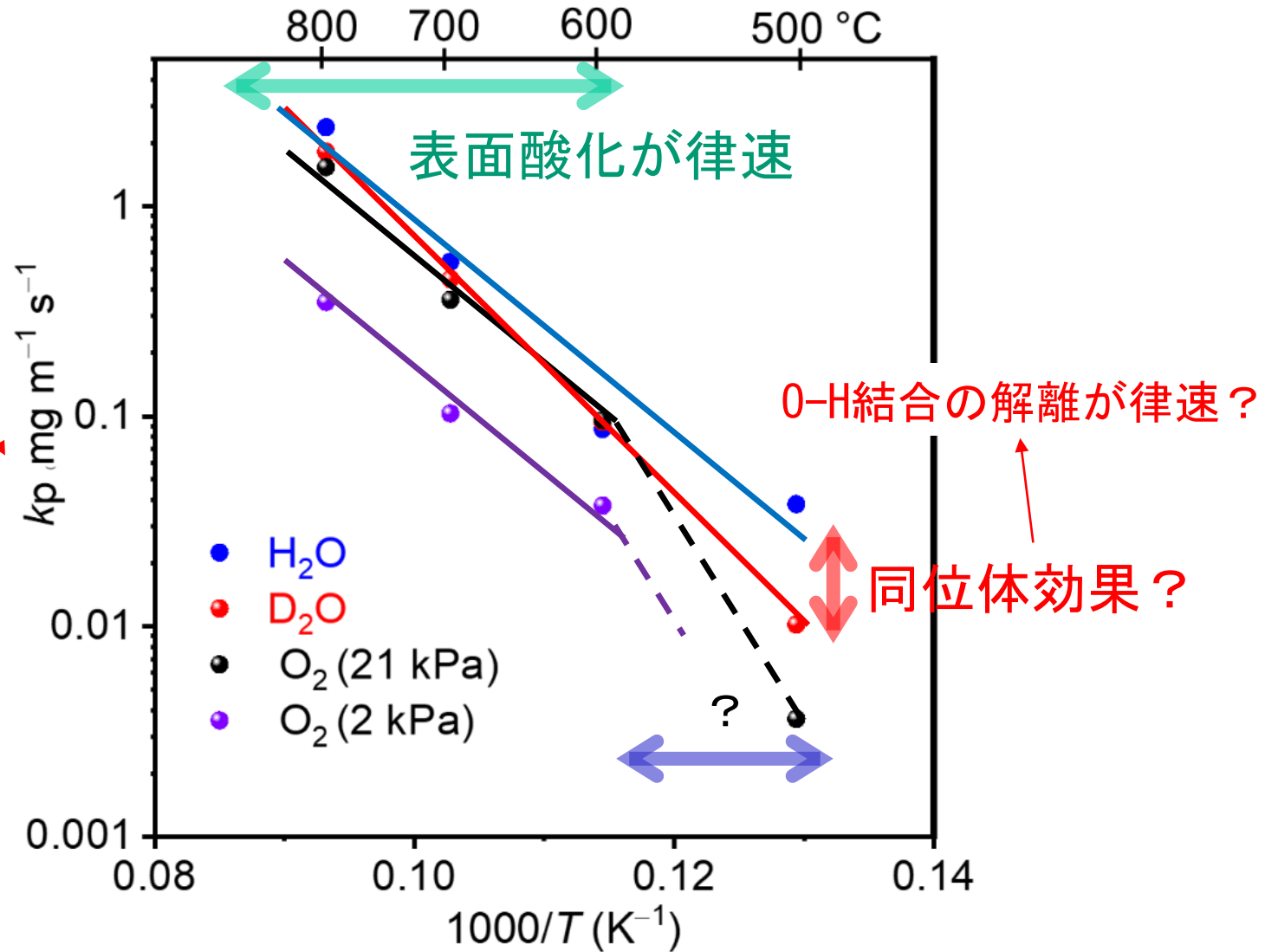
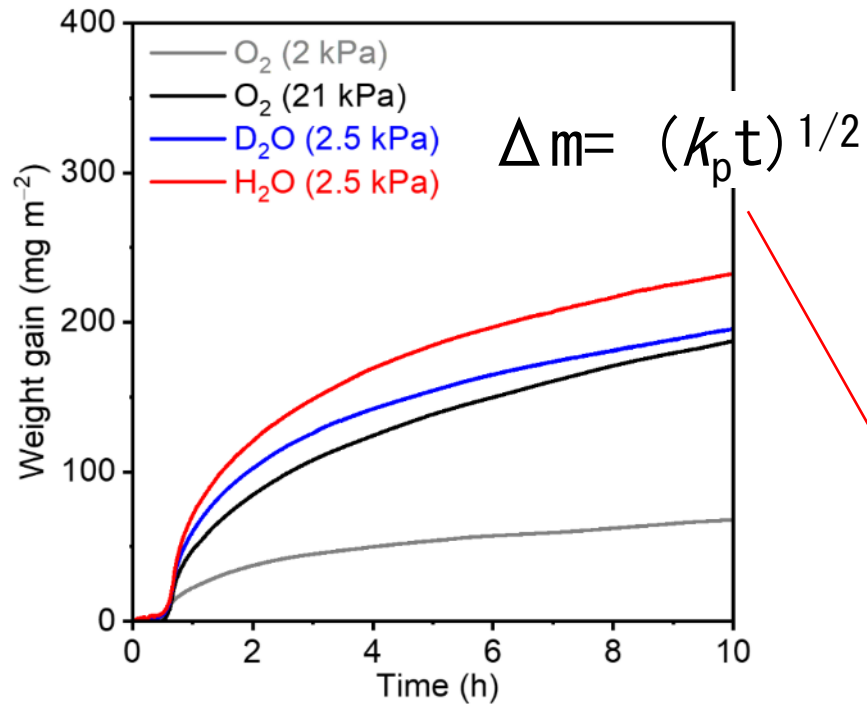


表面の急激な酸化を伴わない水素発生（触媒効果）

酸素による水素発生抑制（ $\leq \frac{1}{10}$ ）
Zryでも

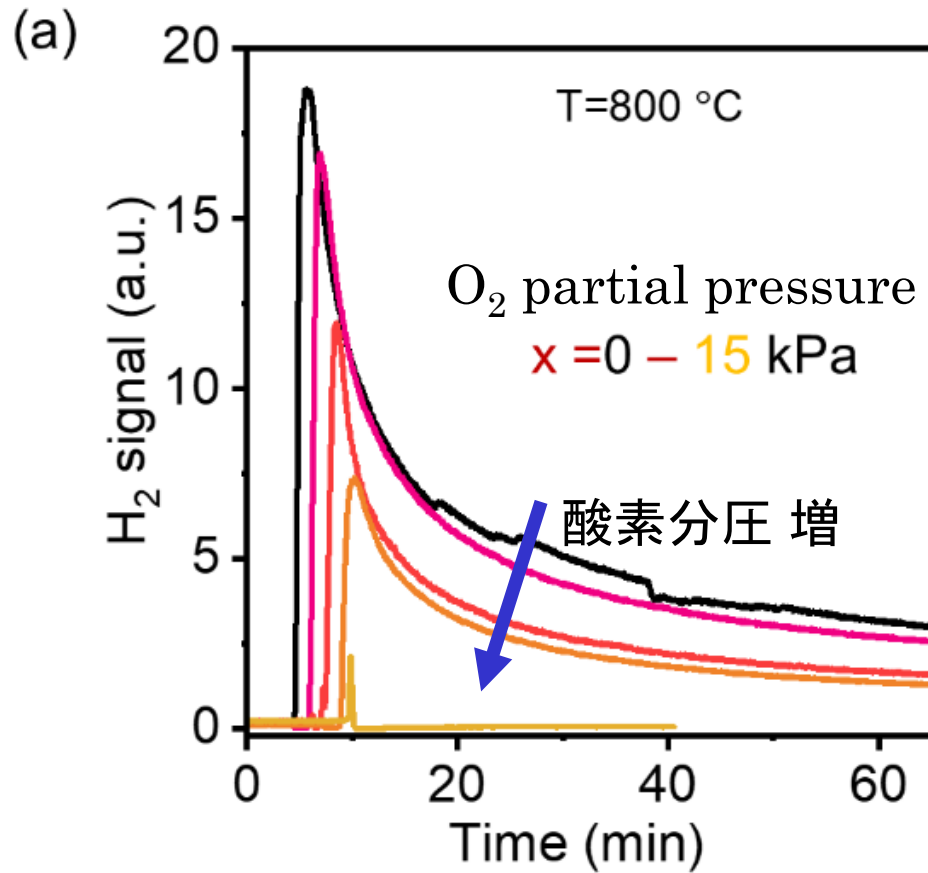


Cr表面酸化の速度論解析

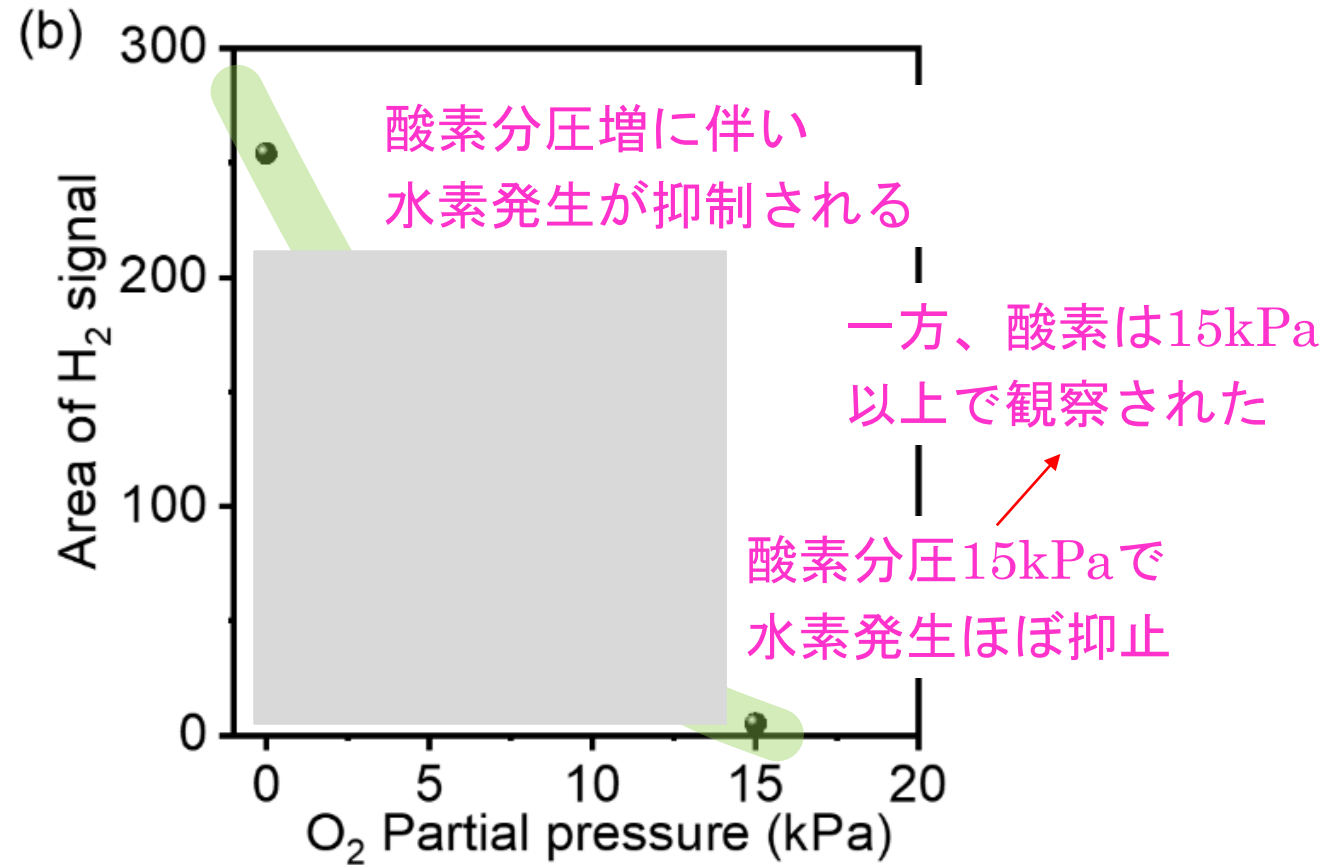




Cr表面での水素発生と抑制因子



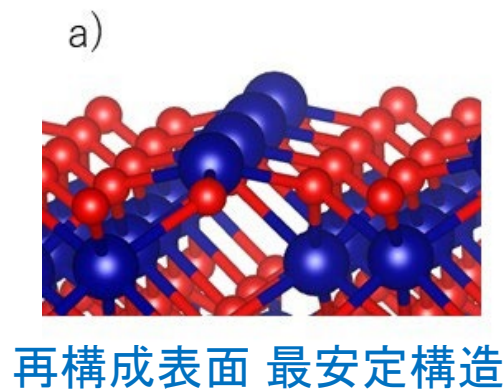
水素発生が暴露初期に限定される
→ 酸化膜による反応抑制機能



ガス中の酸素が水素発生を抑止剤
として機能している

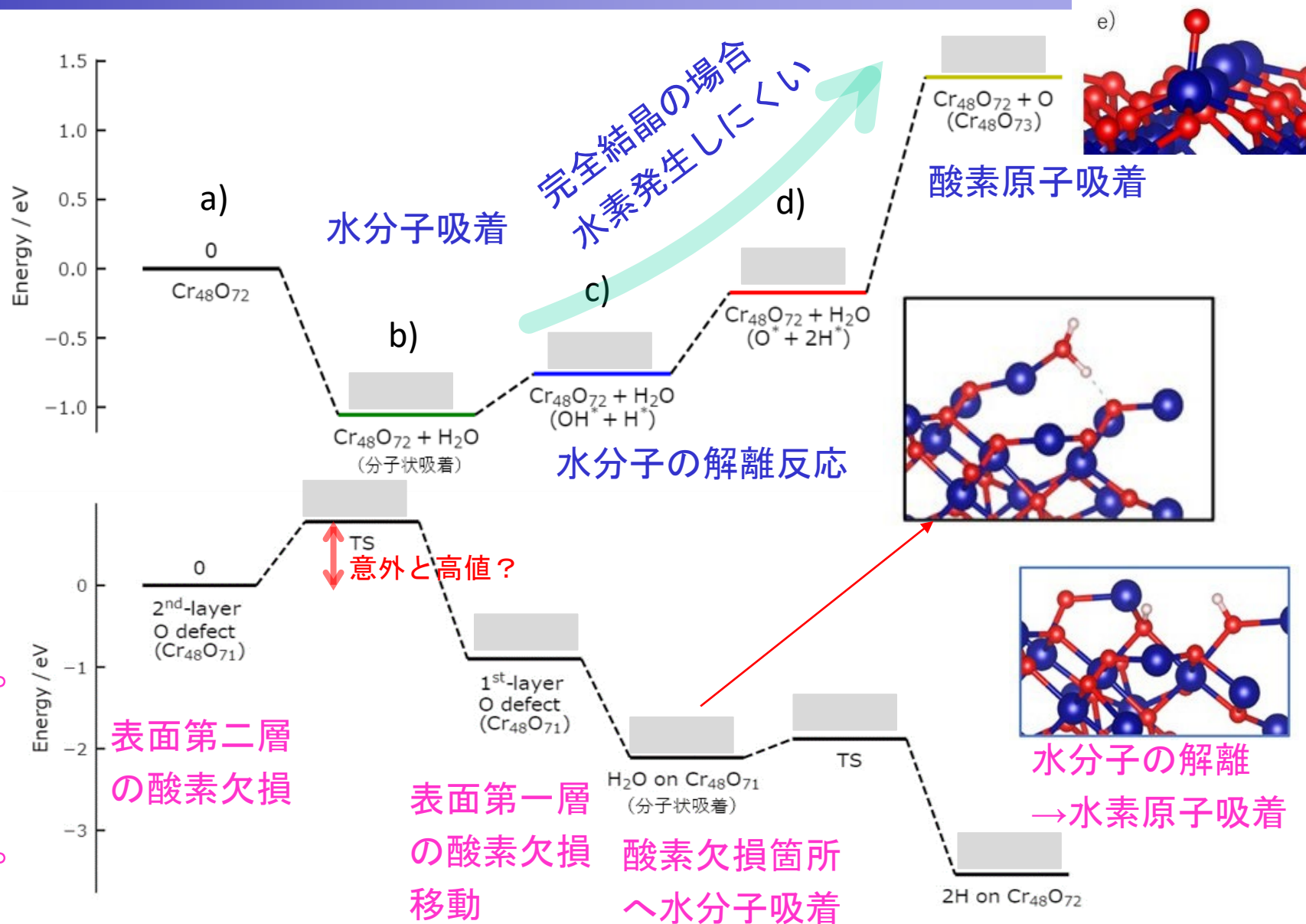


第一原理計算による酸化Cr表面での水素発生



酸素欠損（酸素空孔）は
表面第一層に移動し、
水分子の解離反応を促進する。

完全結晶とは異なる反応過程
であり、今後詳細検討が必要。

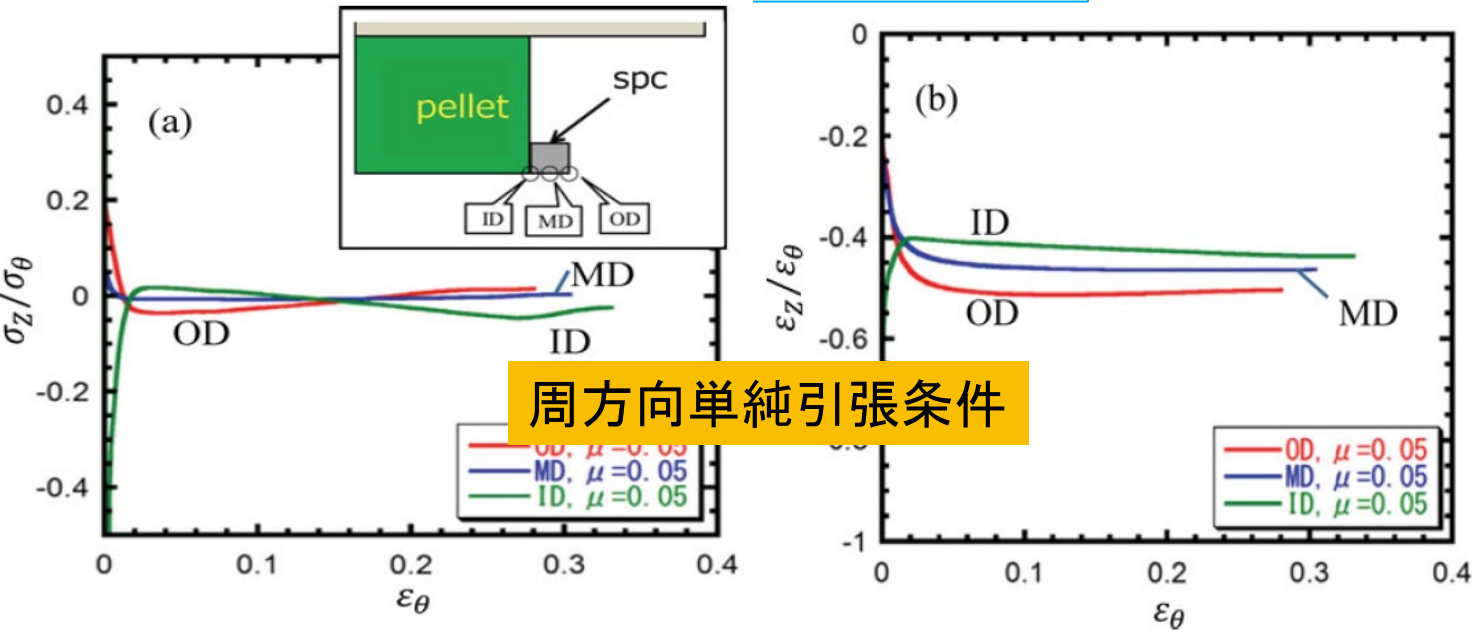
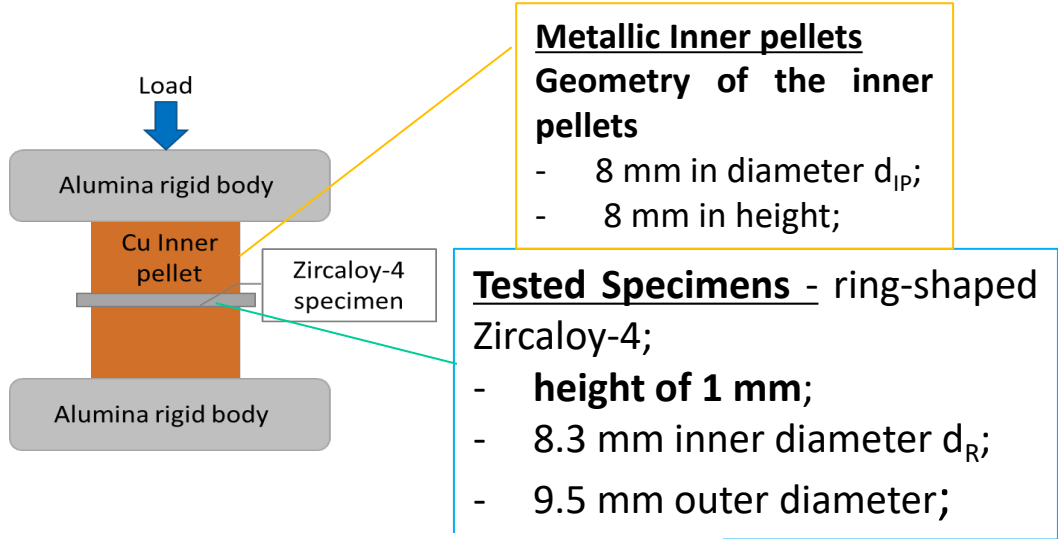




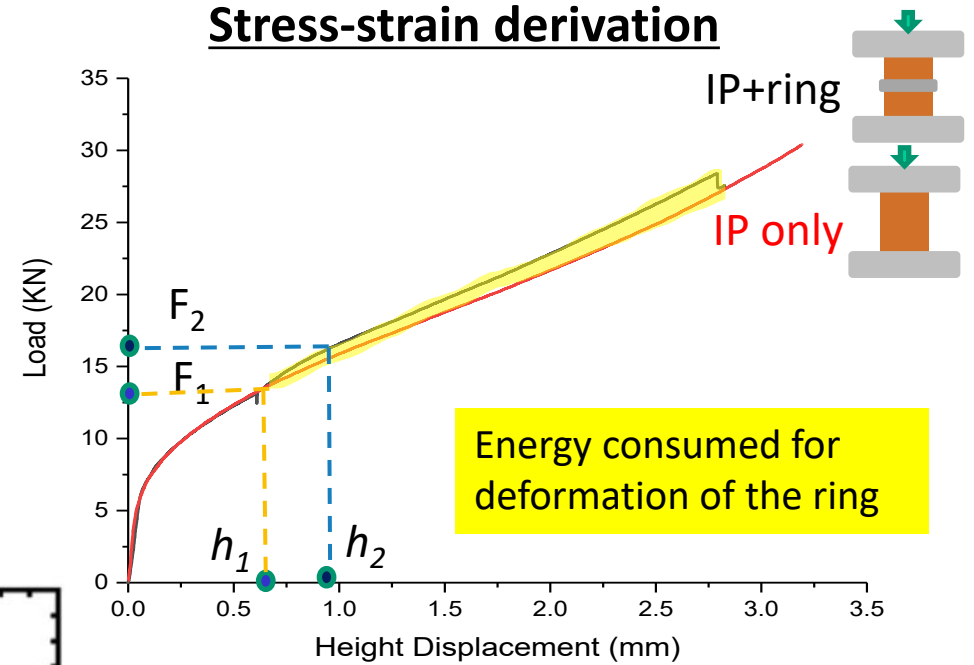
機械的作用



Advanced Expansion Due to Compression Test



Stress-strain derivation



$$\frac{F_1 + F_2}{2} (h_2 - h_1) = E$$

$$\sigma = \frac{(F_1 + F_2) \cdot (h_2 - h_1)}{2V \cdot (\epsilon_2 - \epsilon_1)}$$

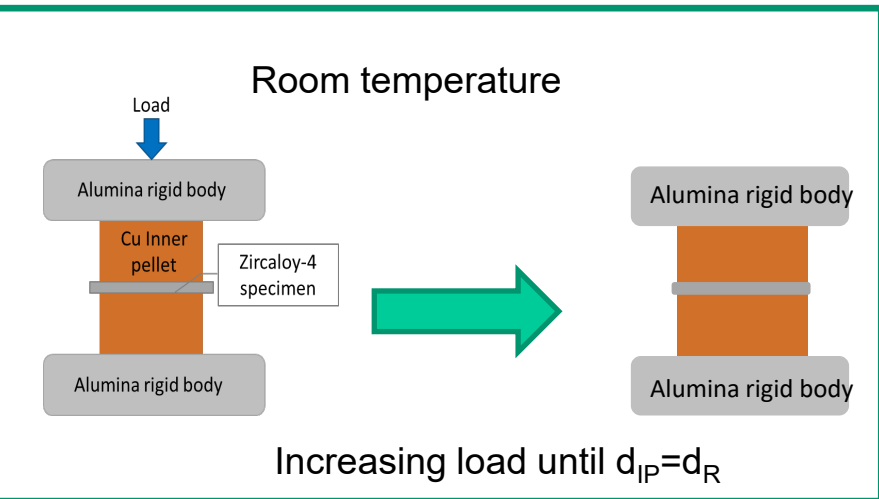
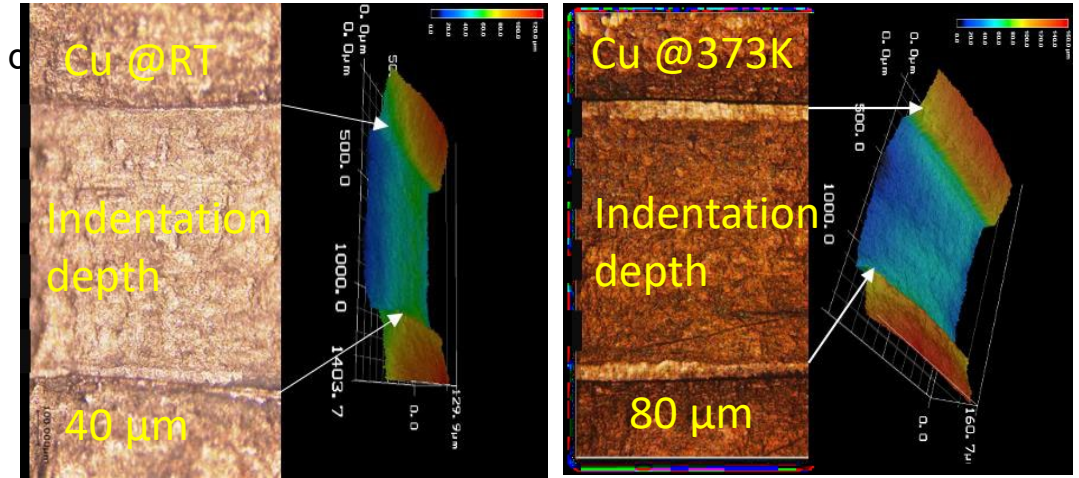


Issues:

- Softening of the Cu inner pellet at high temperatures, leads to hourglass-like deformation
- Max load is limited by the compressive strength of the Alumina plates;

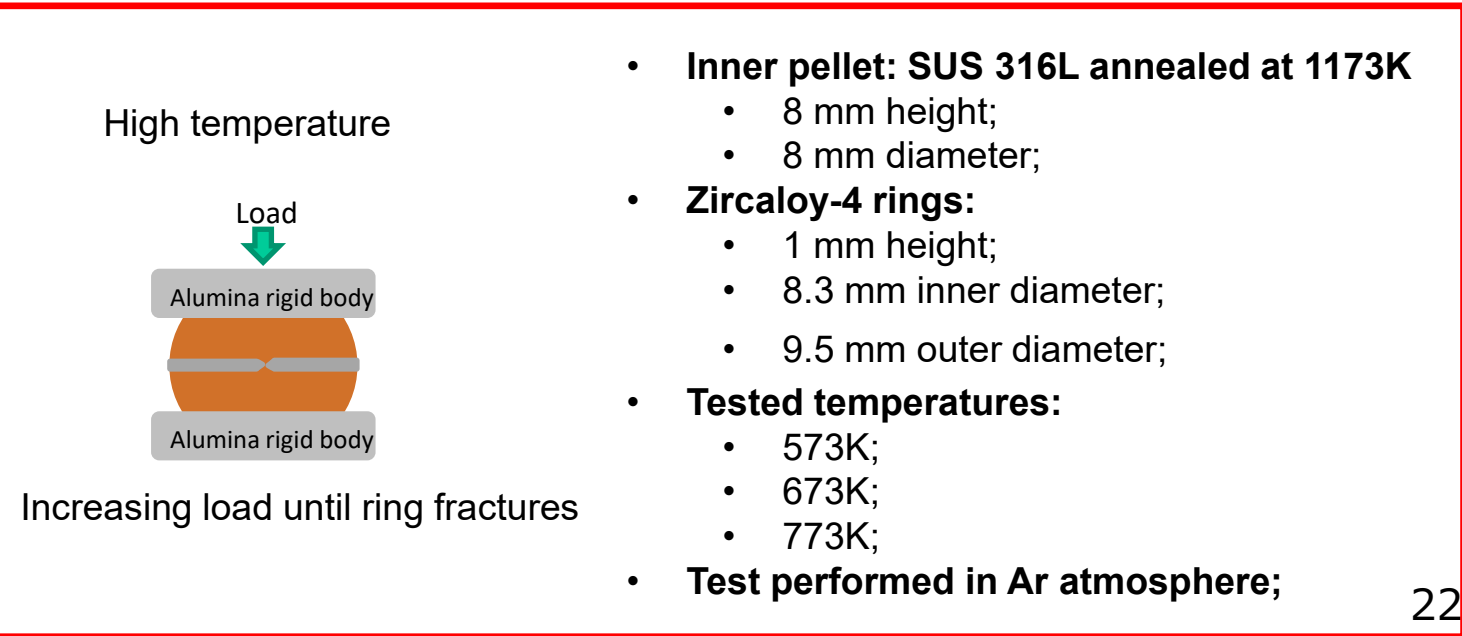
Solutions:

- SUS 316L was used for inner pellet material instead of Cu;
- To soften the material annealing was performed;

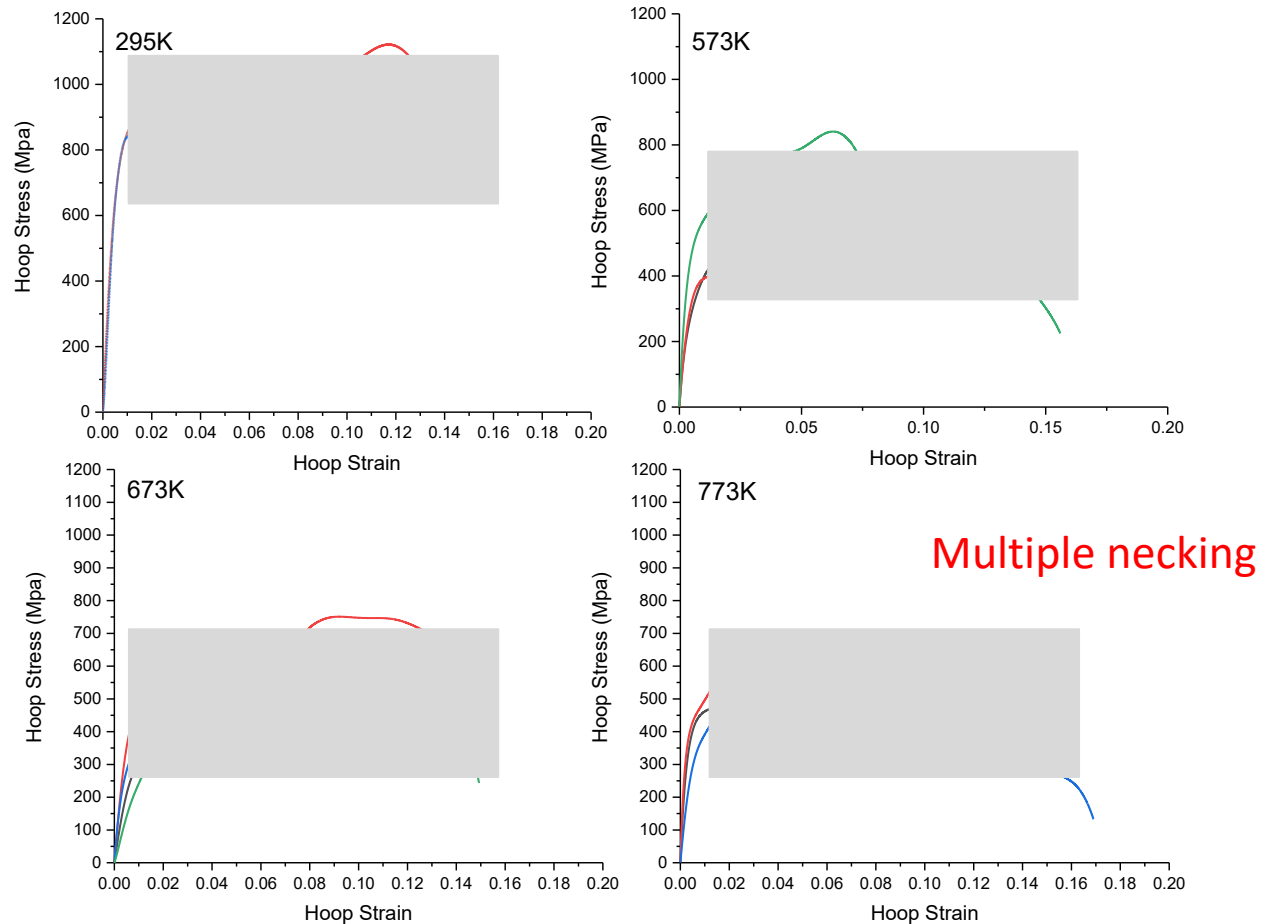


Heating

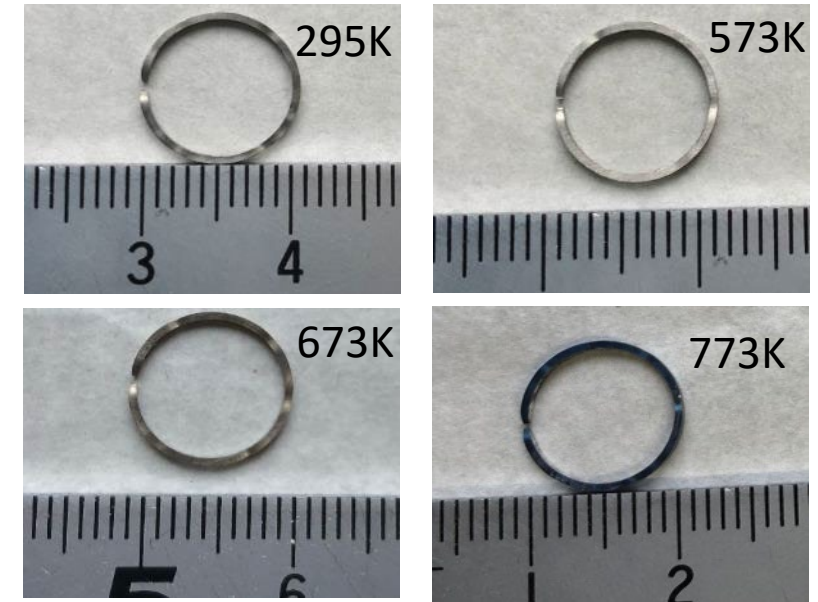
Without loading



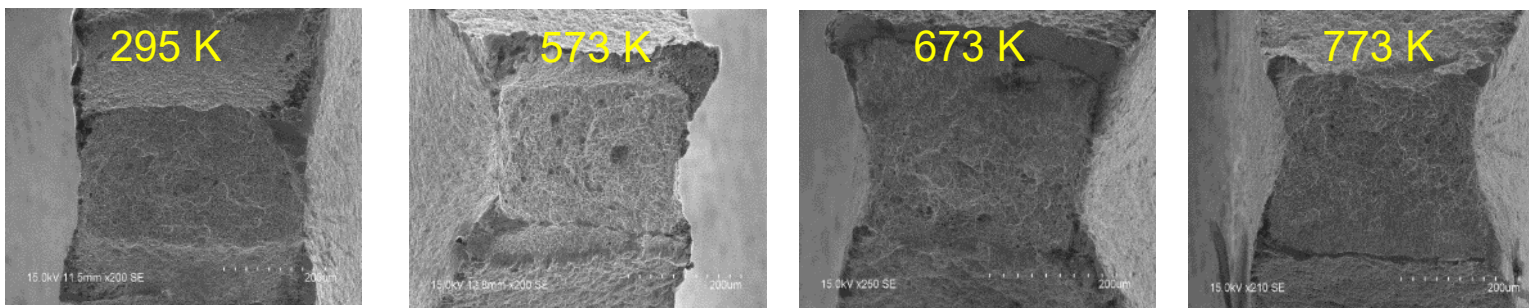
Evaluation of hoop tensile properties in Zircaloy- 4 cladding at elevated temperatures



Zircaloy-4 ring samples



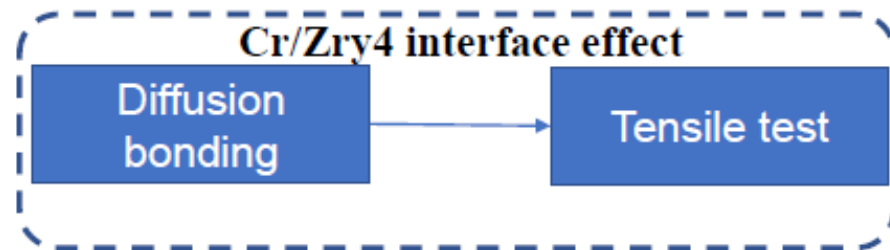
- Multiple necking observed in all tested samples;
- Necking becomes deeper with increasing temperature;
- Samples started oxidizing at temperature above 673K;
- Ductile fracture observed for each tested temperature;
- Dimple size and depth increase with temperature;





拡散接合材の強度計測

1. experiment

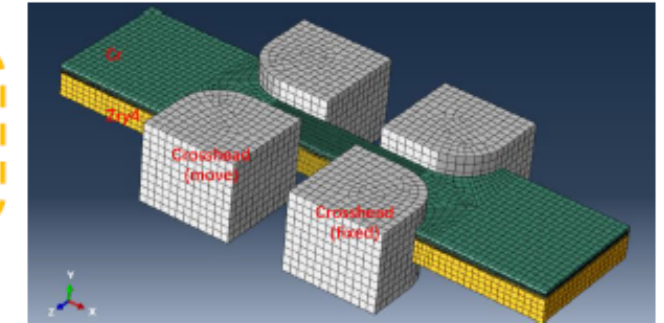


Tensile test

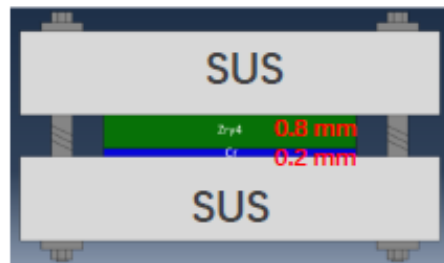
2. Numerical modelling



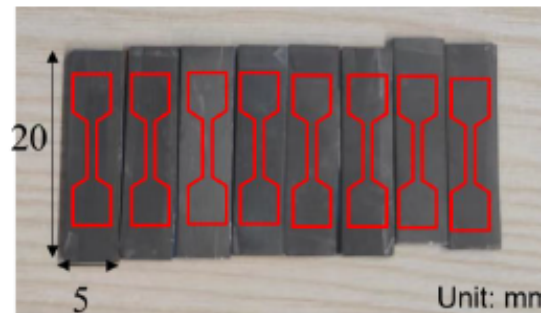
Model built in Abaqus



Diffusion bonding set-up:

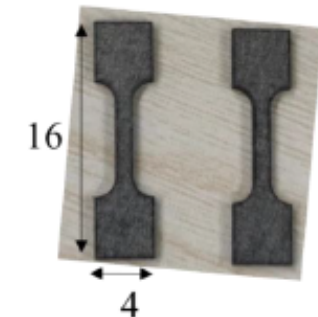
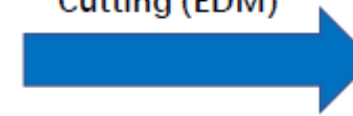


800°C (or 1000°C), 4h
vacuum



As-bonded Cr/Zry4 bilayers

Cutting (EDM)



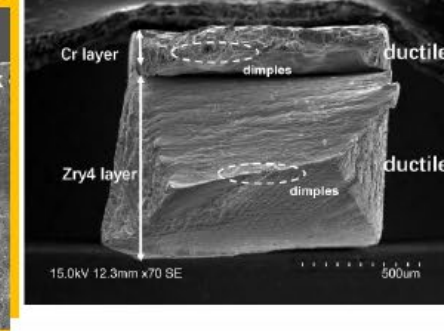
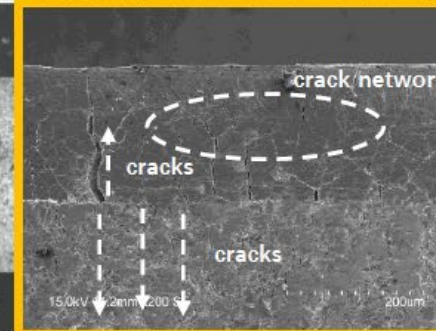
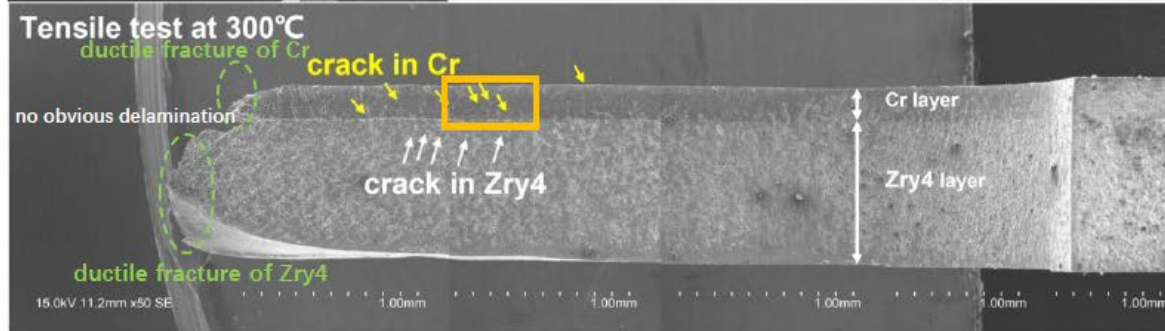
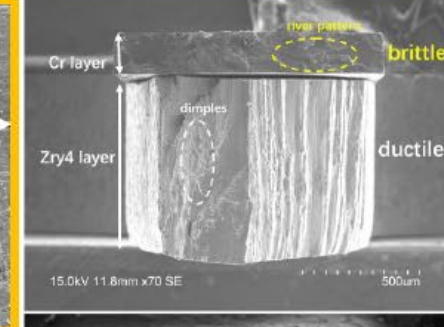
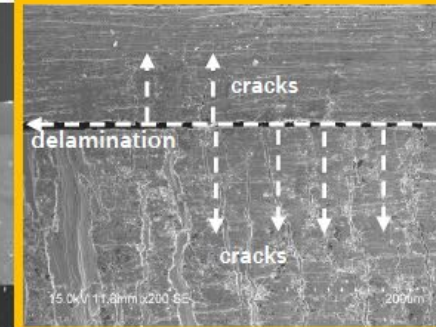
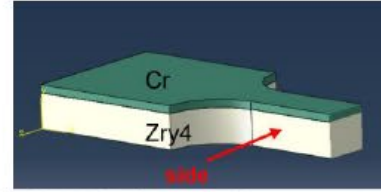
Tensile specimen of Cr/Zry4 bilayers

Stress evolution of Cr/Zry4 bilayer, help explain the experimental results of diffusion-bonded specimen.

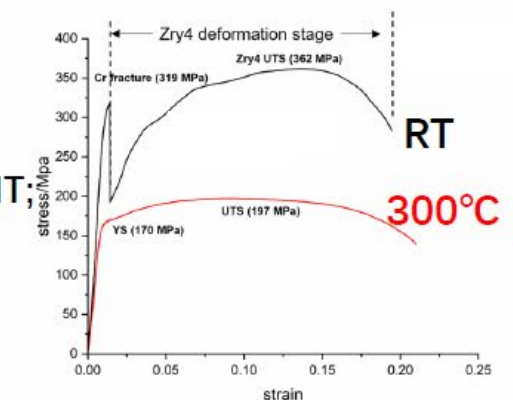


Fractography of Cr-Zry4 bilayer

800°C-bonded

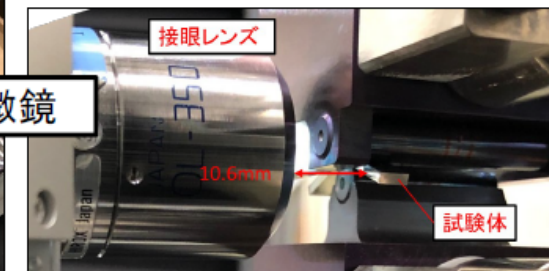
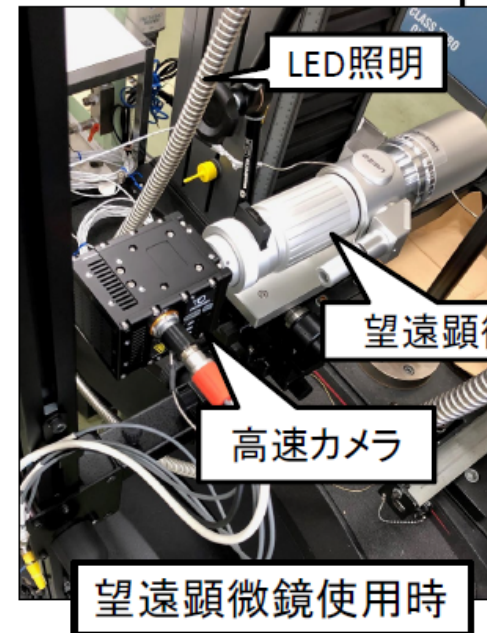
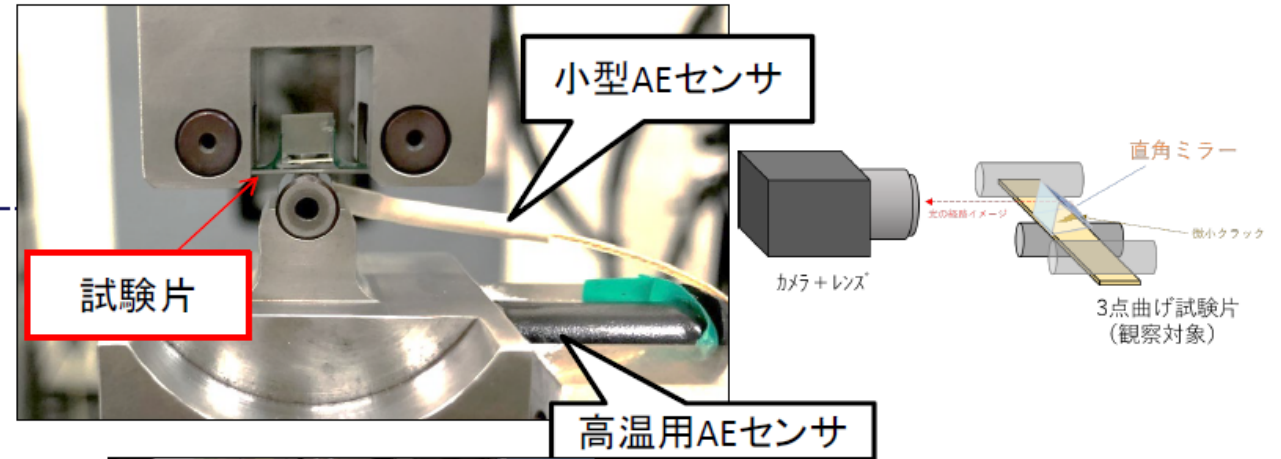
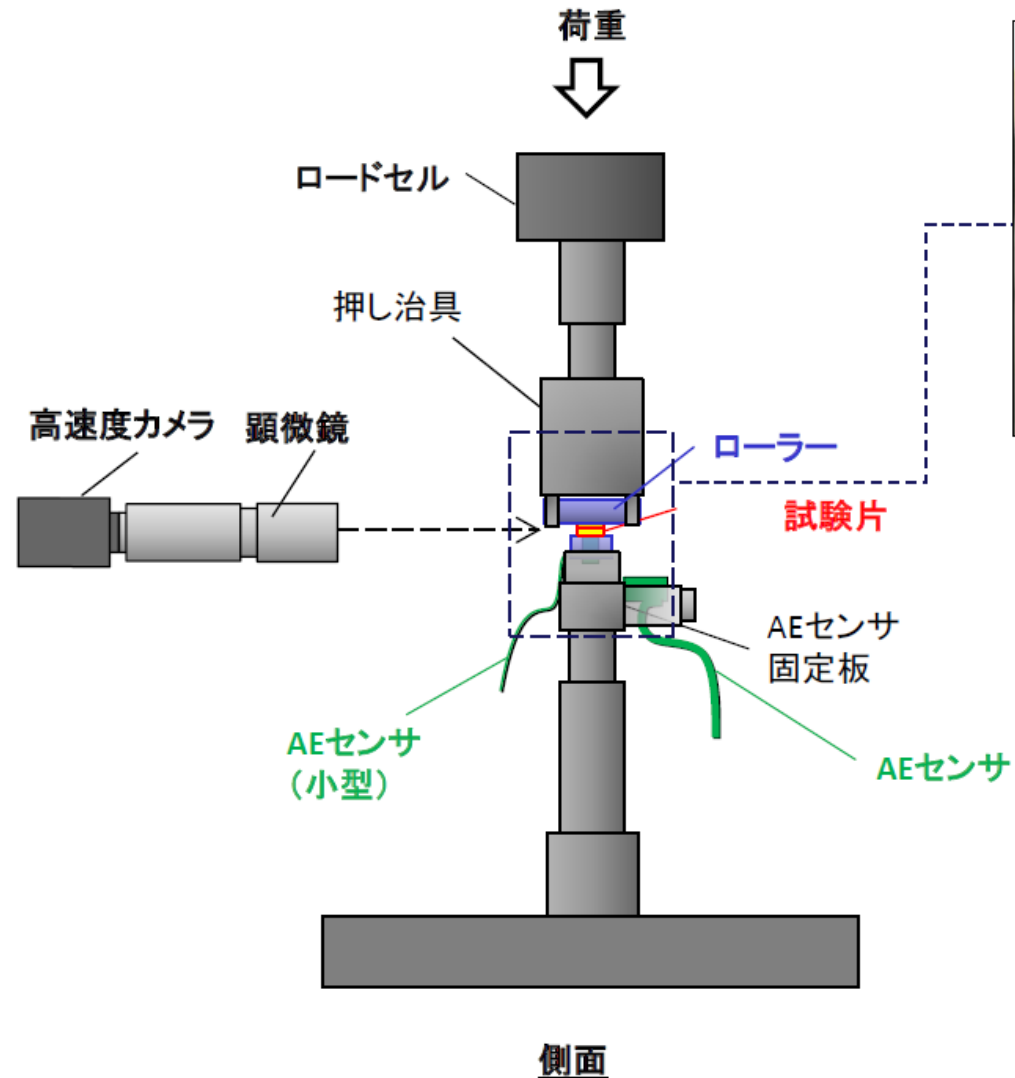


- Two-stage fracture at RT, while no mechanical response at HT;
- Clean delamination without no large peeling-offs at RT, while no obvious delamination at HT;
- A **strong adhesion** is obtained for 800°C-bonded Cr/Zry4 bilayer.





試験装置



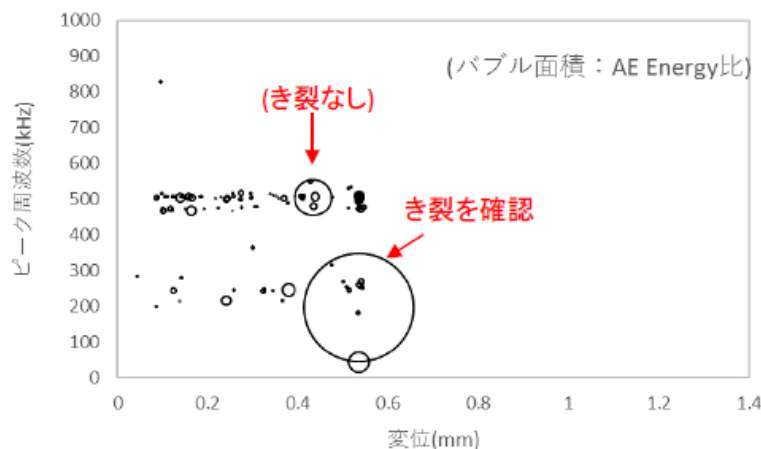
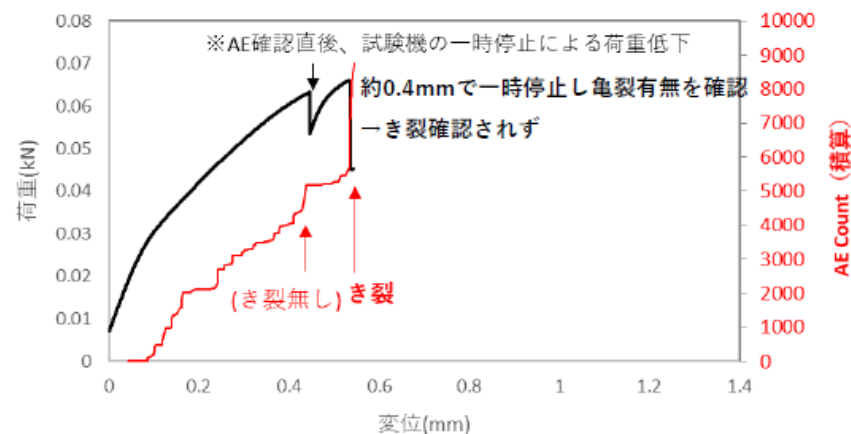


試験結果

NCC

拡散接合材 Cr 0.2mm + Zry 0.5mm

撮影速度: 1/10000sec (近接顕微鏡)



- 母材変化部
- き裂の発生と対応しないAE信号を取得
→剥離に対応する信号と推定
- 1/10000sec以下の短時間で母材状態の変化が発生
→荷重変化から、き裂と同時と推定
→同時刻にAE Energyの強いAE信号を確認

図 上: 荷重-変位 + AE活動 下: AE信号-変位



物理的作用

背景と目的②

モチベーション

重畳影響(核変換H, He + 照射)下での
照射誘起非晶質化反応は？

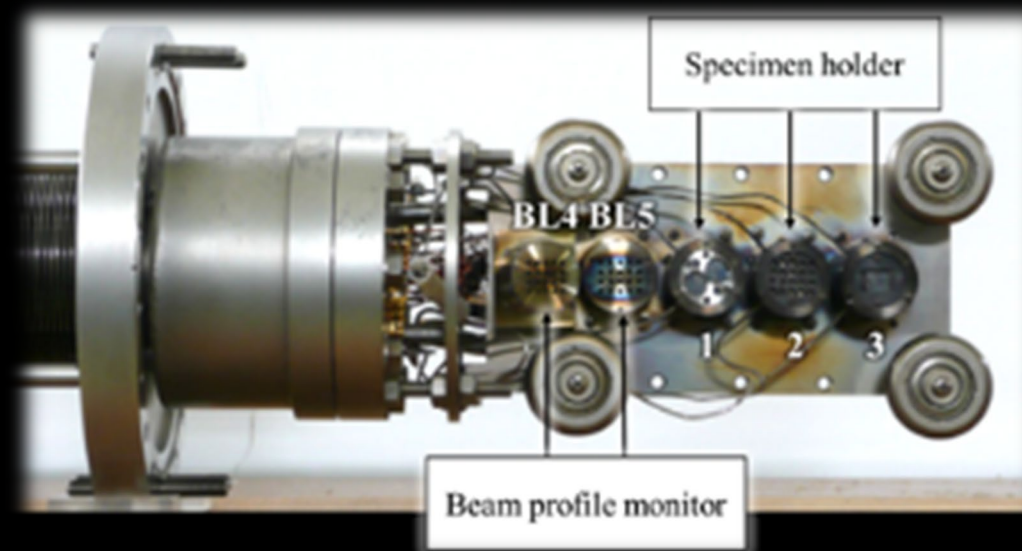
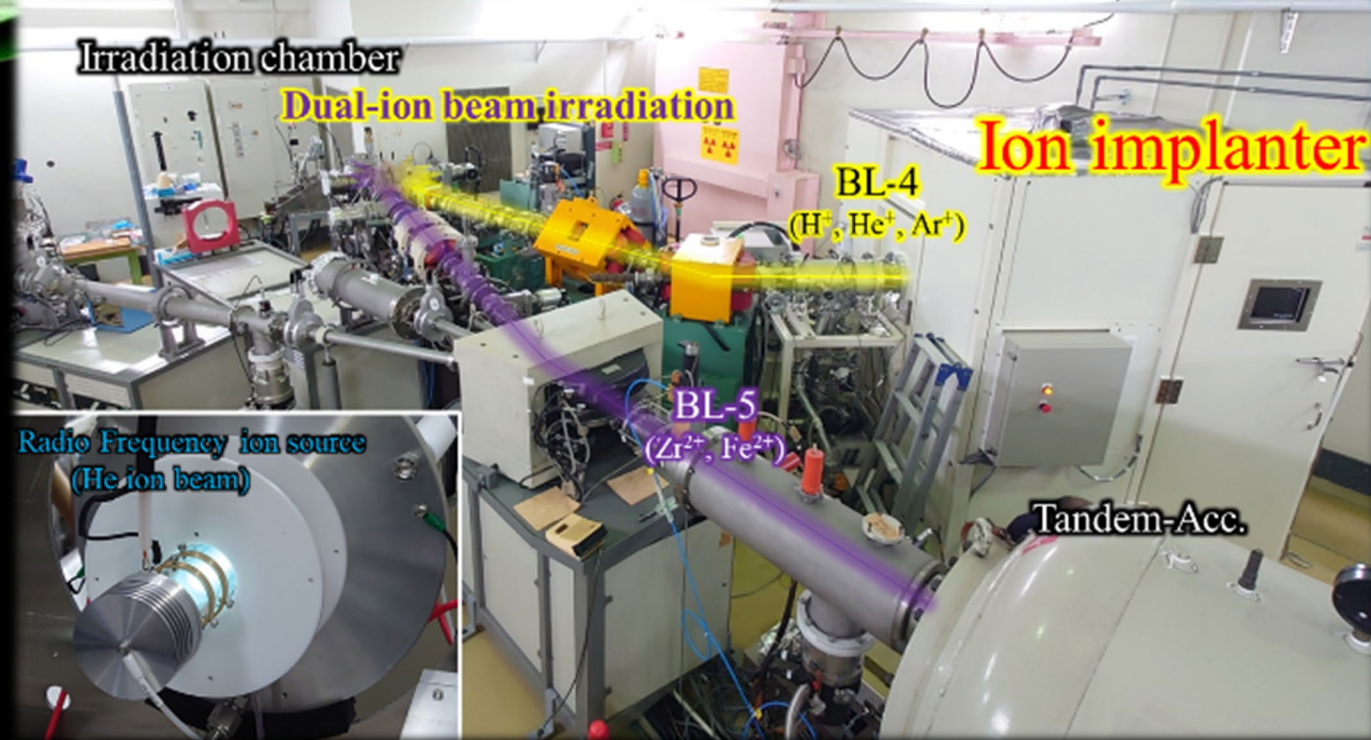
マルチイオンビーム照射 は、有効な評価手段

震災でバンデグラフが被災、廃棄の決定.

HITのタンデトロンビームラインに
イオン注入装置からのビームラインを連結、
二重照射設備を更新(再稼働)を実施した.

<目的>

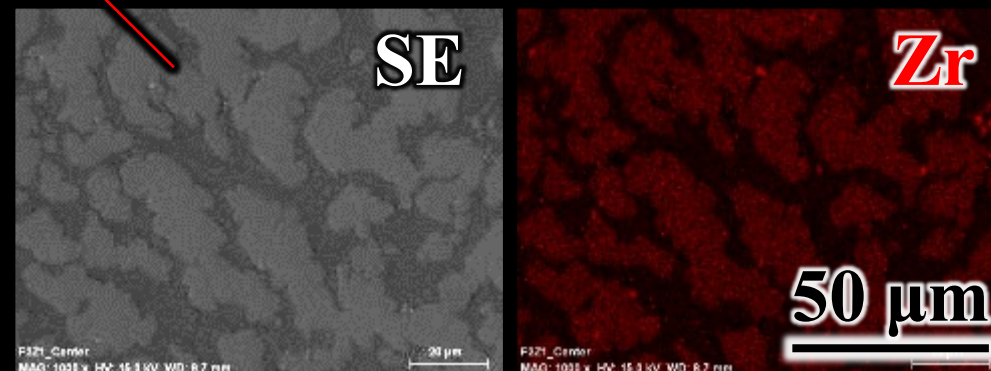
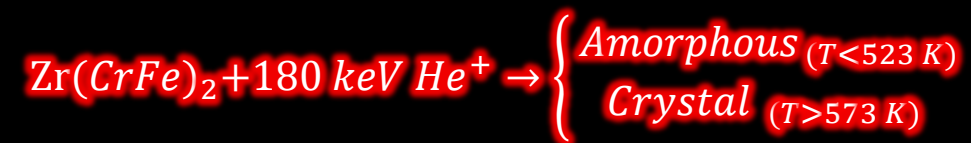
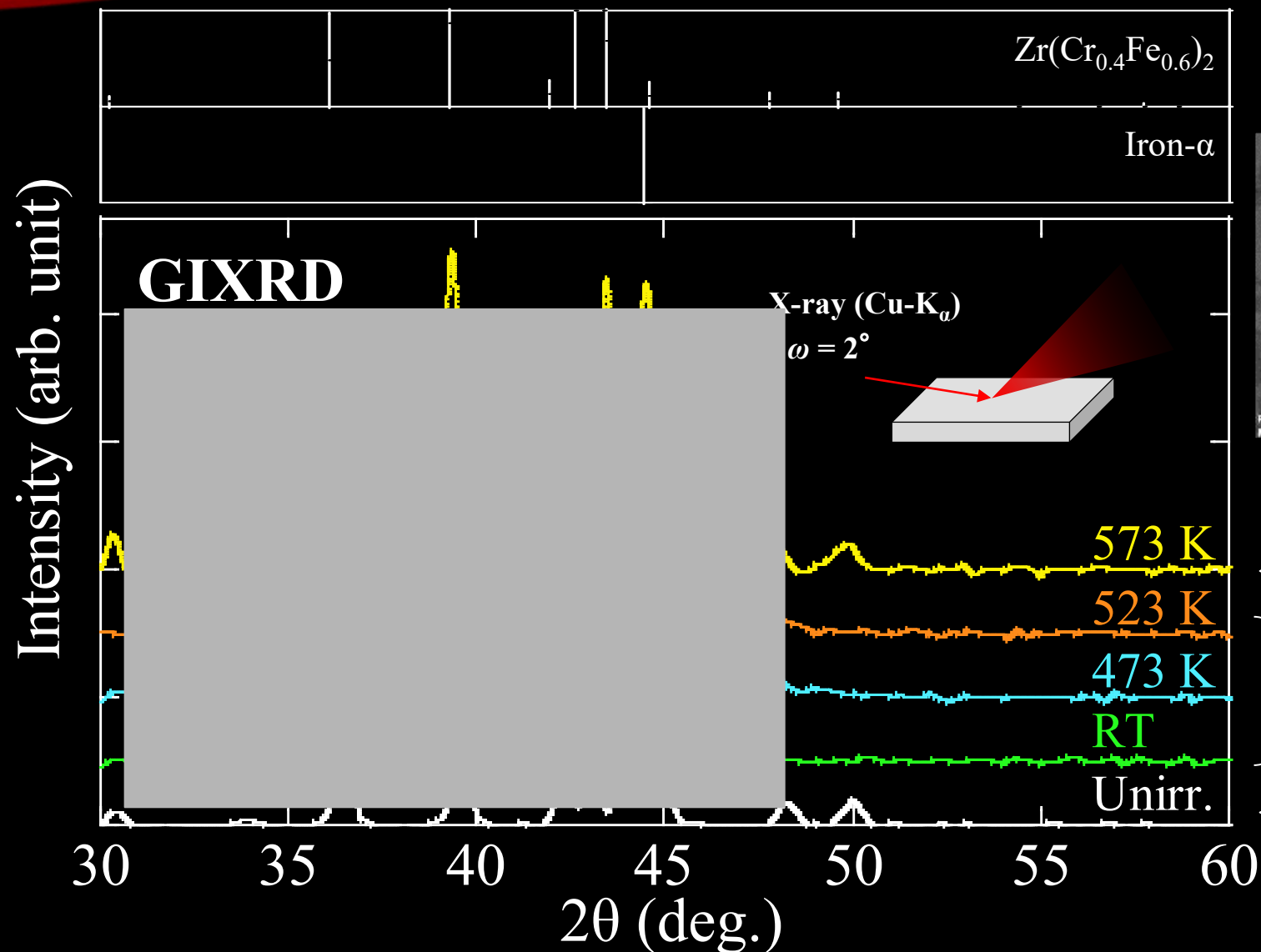
- イオン注入装置の照射性能確認
- 金属間化合物相の
照射誘起相変態挙動を明らかにする



結果 : $\text{Zr}(\text{Cr,Fe})_2$ ラーベス相の照射誘発非晶質化 (RIA)

GIXRD *Grazing Incident X-ray Diffraction*

$\text{Cu-K}\alpha$, $\omega = 2 \text{ deg.}$, $\text{Step} = 0.001 \text{ deg.}$

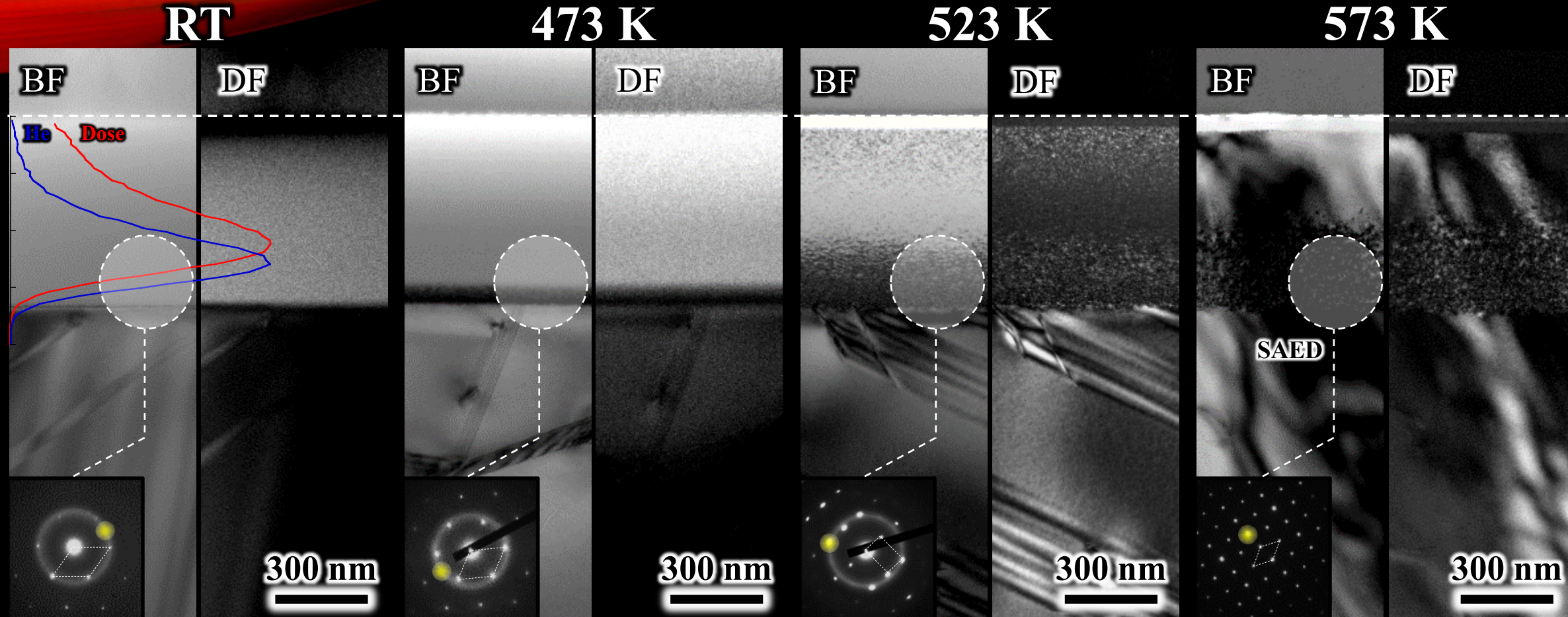


← 未照射材と同様

ピークが消失、非晶質ピークが出現
(母相 Iron- α は、変化なし)

← $\text{Zr}(\text{Cr,Fe})_2$ からの回折ピーク

結果：縦断面TEM観察



Zr(Cr,Fe)₂の照射誘起非晶質化(RIA)を確認

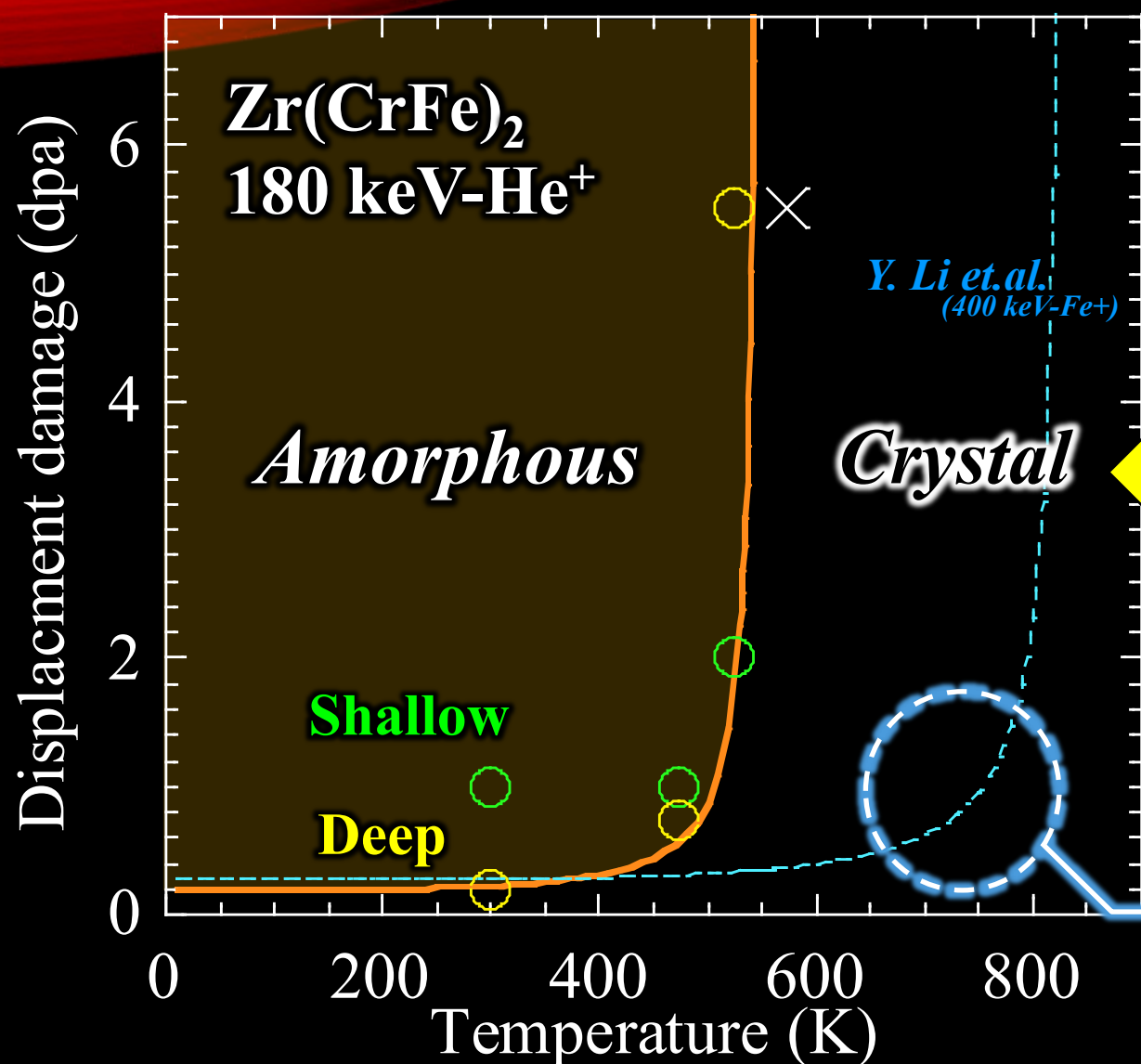
RT & 473 K : ダメージ領域の概ね全域

523 K : 中央領域のみ(上端/下端部は結晶性を維持)

573 K : RIAは確認されない

ダメージピークの前方(Shallow)、
後方(Deep)でのRIA閾照射量を評価。

結果：照射温度、照射量依存性



$$\text{Direct-impact model : } D_C = \frac{D_0}{1 - \exp\left[\left(\frac{E_a}{k}\right)\left(\frac{1}{T_C} - \frac{1}{T}\right)\right]}$$

- D_C : Critical dose for amorphization
- D_0 : " " extrapolated at $T = 0$ K
- E_a : Activation energy for the dynamic annealing of a cascade
- T_C : Critical temperature for amorphization
- T : Irradiation temperature
- k : Boltzmann constant.

$$T_C : \sim 548 \text{ K}$$

$$E_a : \sim 0.13 \text{ eV} \text{ (} D_0 \text{ を } 0.2 \text{ dpa と仮定)}$$

Liらの結果も $E_a < 1$ eVと見積もっている。

“弾き出し+回復” プロセスでは説明できない。
 E_a に物理的意味はない。

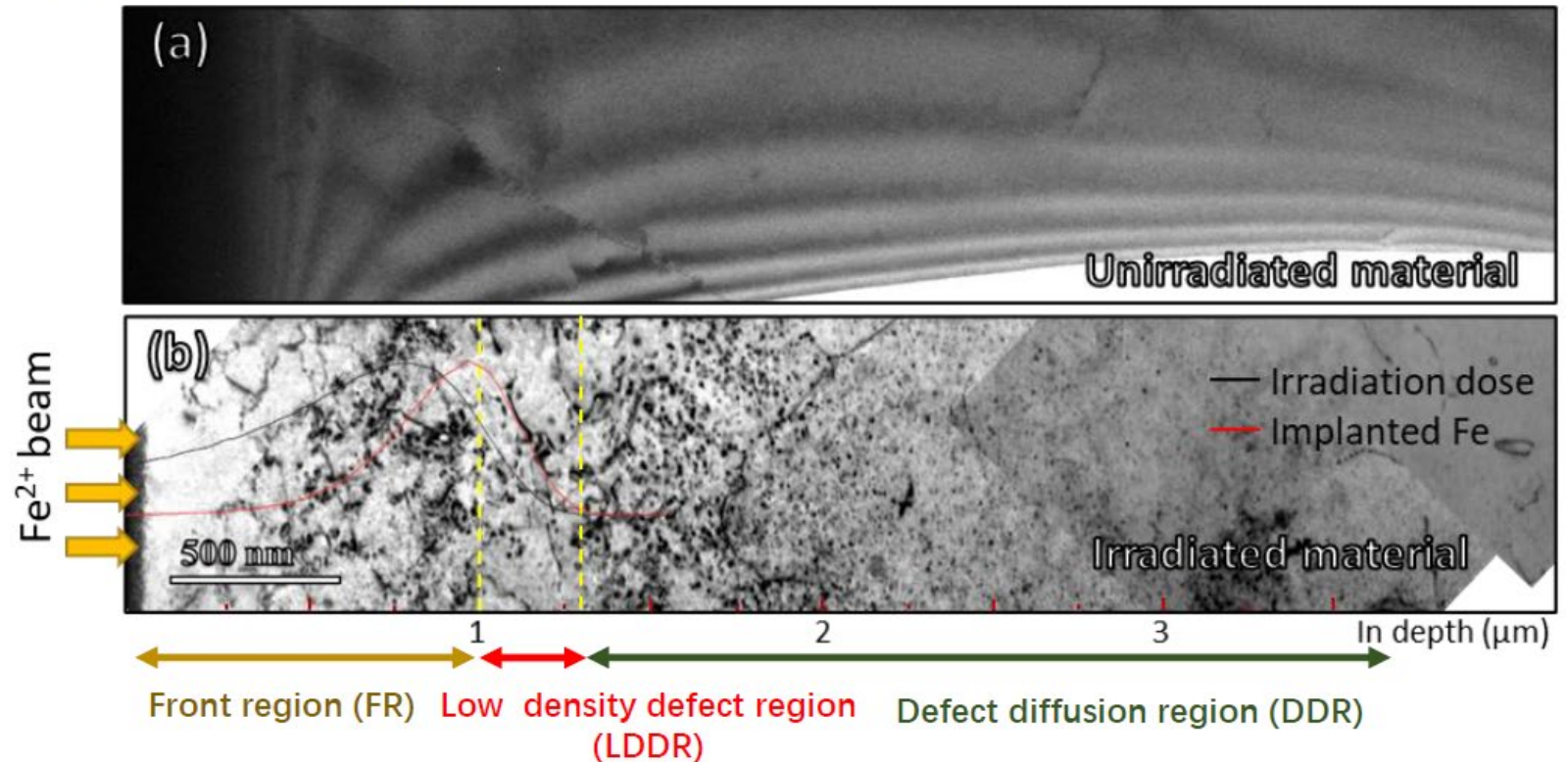
Q : 緩やかな立ち上がりが意味する物理描像は？

- Ref. [1] Li et.al., NIMB (2022)
[2] Weber et.al., NIMB (2000)
[3] Wang et.al., Phys. Rev. (2000)



照射損傷の観察、イオン注入影響確認

- The defect distribution depth $\gg$ the Fe penetration depth $>$ the irradiation damage depth.
- According to the defect distribution:
 - I. Front region
 - II. Low density defect region
 - III. Defect distribution region
- Voids only formed in the irradiated region.





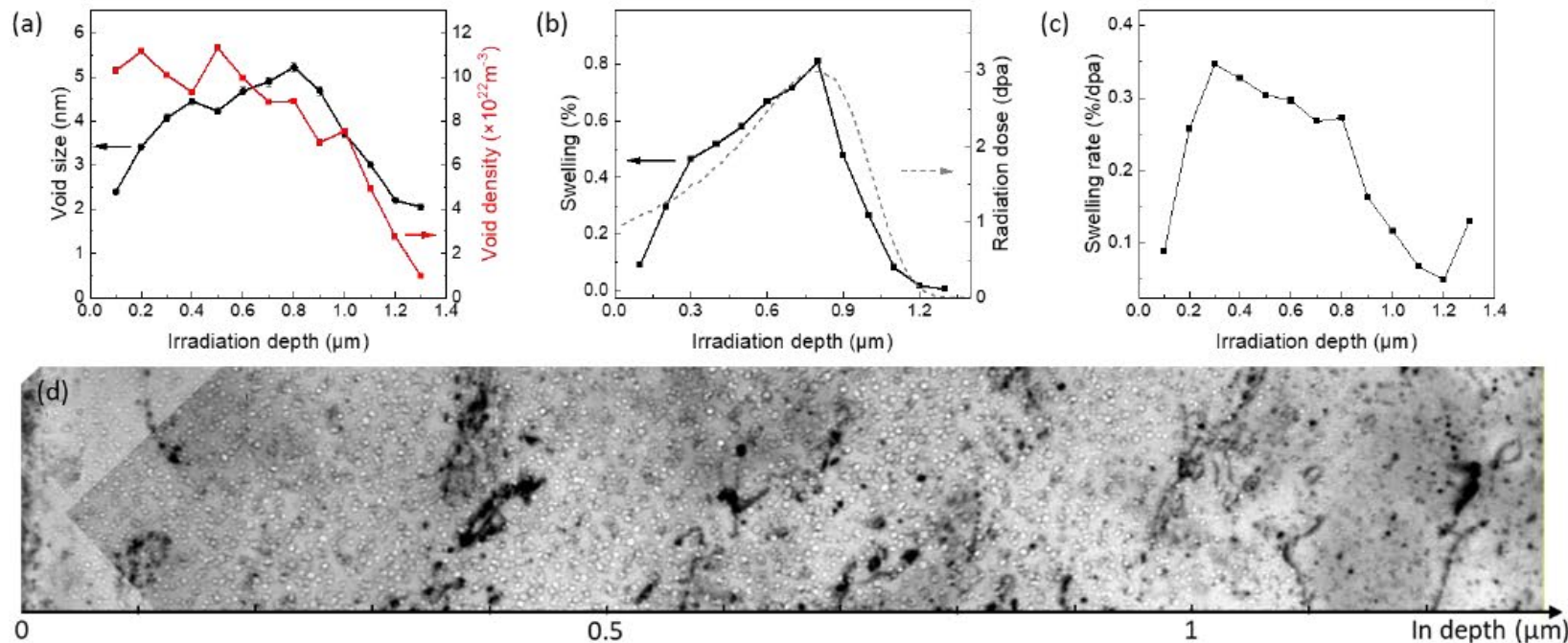
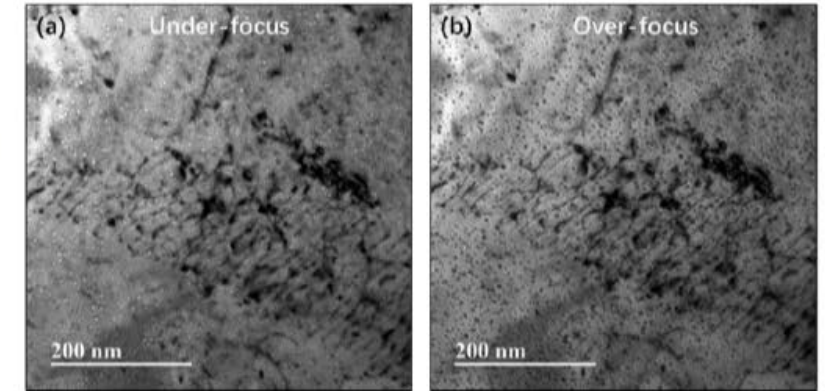
ボイドスウェリングの確認

Irradiation: 2.8 MeV Fe²⁺ @ 550±2 °C to 3 dpa

Sample preparation: FIB + Flash polishing

TEM observation

- The swelling was quantified based on the void size and density.
- In general, the swelling matches the SRIM simulated irradiation doses distribution.
- The swelling rate close to the peak region is around 0.3%/dpa.
- Grain boundaries can act as an effective sink for point defects.

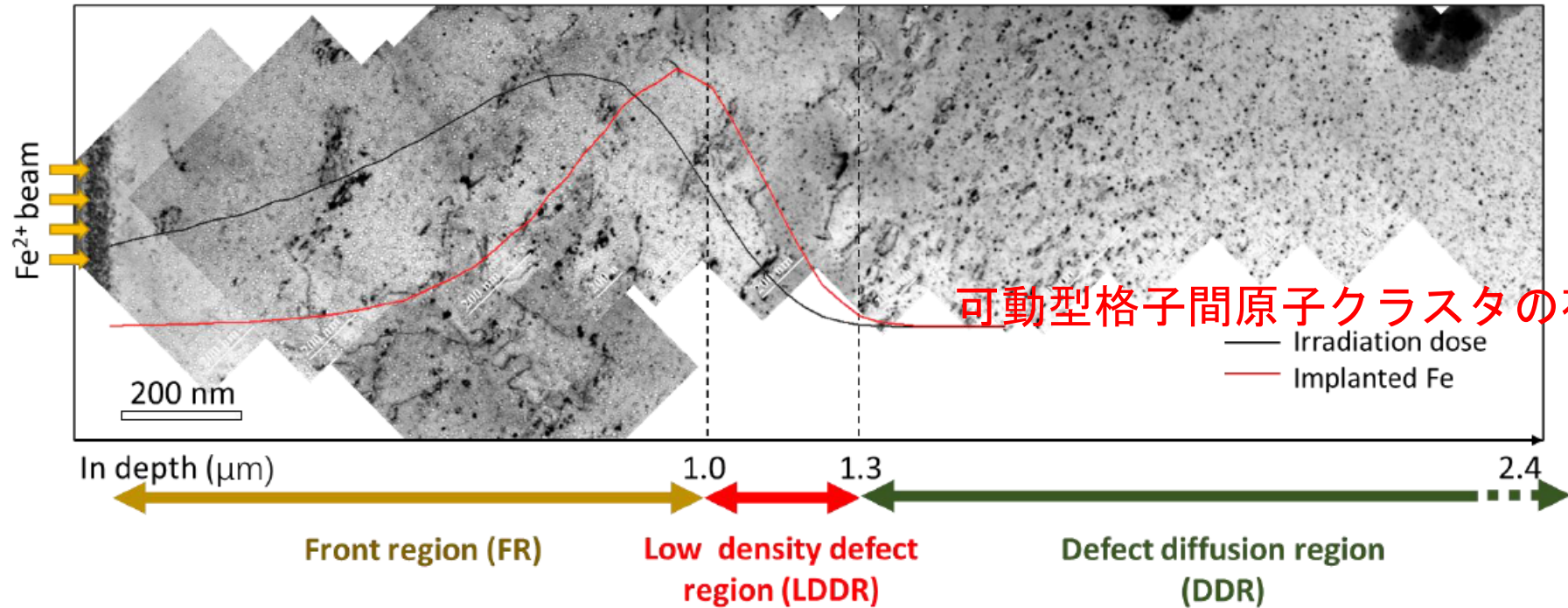




照射欠陥（転位ループ）の観察

- All the loops $b=1/2 \langle 111 \rangle$ and mostly are pure edge loops.
- The interstitial loop ratio increases with irradiation depth. (analyzed ~ 100 loops)

Region	FR	LDDR	DDR	
In depth (nm)	0-1000	1000-1300	1300-1600	2200-2400
I-loop ratio				



可動型格子間原子クラスタの存在の確認



まとめ

- Cr被覆ジルカロイの開発のため、体系的に課題を設定し、材料開発、機械的作用、物理的作用、化学的作用を総合的に研究する体制を構築した。
- 実験的理論的研究の進捗により被覆材候補合金の材料開発が進行中である。
- 化学的作用について、Cr被覆による腐食抑制と水素発生抑制の確認とその機構解明を行った。
- Crでは酸化と水素発生の両方が大幅に抑制されることを確認した。
- 一方で、金属表面における触媒反応に起因した（酸化を伴わない）水素発生については、Cr、Zry双方で確認された。
- 機械試験法として、管形状材料の機械的性質評価法開発、その場機械試験法開発を進めており、今後Cr/Zry接合材へ本格的用が進む。
- 物理的作用に関し、Cr中の照射損傷の形成の詳細解析が進んでいる。